



A data-driven review of thermoelectric materials: Performance and resource considerations

Citation

Gaultois, Michael W., Taylor D. Sparks, Christopher K. H. Borg, Ram Seshadri, William D. Bonificio, and David R. Clarke. Forthcoming. Data-driven review of thermoelectric materials: Performance and resource considerations. *Chemistry of Materials*.

Published Version

doi:10.1021/cm400893e

Permanent link

<http://nrs.harvard.edu/urn-3:HUL.InstRepos:10665230>

Terms of Use

This article was downloaded from Harvard University's DASH repository, and is made available under the terms and conditions applicable to Other Posted Material, as set forth at <http://nrs.harvard.edu/urn-3:HUL.InstRepos:dash.current.terms-of-use#LAA>

Share Your Story

The Harvard community has made this article openly available.
Please share how this access benefits you. [Submit a story](#).

[Accessibility](#)

A data-driven review of thermoelectric materials: Performance and resource considerations

Michael W. Gaultois,^{*,†,‡} Taylor D. Sparks,^{*,‡} Christopher K. H. Borg,[‡]

Ram Seshadri,^{*,¶,†,‡} William D. Bonificio,[§] and David R. Clarke[§]

*Department of Chemistry and Biochemistry, University of California, Santa Barbara, CA
93106, Materials Research Laboratory, University of California, Santa Barbara, CA 93106,
Materials Department, University of California, Santa Barbara, CA 93106, and School of
Engineering and Applied Sciences, Harvard University, 29 Oxford Street, Cambridge, MA
02138*

E-mail: mgaultois@mrl.ucsb.edu; sparks@mrl.ucsb.edu; seshadri@mrl.ucsb.edu

*To whom correspondence should be addressed

[†]Department of Chemistry and Biochemistry, UCSB

[‡]Materials Research Laboratory, UCSB

[¶]Materials Department, UCSB

[§]School of Engineering and Applied Sciences, Harvard University

Abstract

In this review, we describe the creation of a large database of thermoelectric materials prepared by abstracting information from over 100 publications. The database has over 18 000 data points from multiple classes of compounds, whose relevant properties have been measured at several temperatures. Appropriate visualization of the data immediately allows certain insights to be gained with regard to the property space of plausible thermoelectric materials. Of particular note is that any candidate material needs to display an electrical resistivity value that is close to $1 \text{ m}\Omega \text{ cm}$ at 300 K, *i.e.*, samples should be significantly more conductive than the Mott minimum metallic conductivity. The Herfindahl-Hirschman index, a commonly accepted measure of market concentration, has been calculated from geological data (known elemental reserves) and geopolitical data (elemental production) for much of the periodic table. The visualization strategy employed here allows rapid sorting of thermoelectric compositions with respect to important issues of elemental scarcity and supply risk.

Keywords: Thermoelectrics, datamining, Herfindahl-Hirschman Index, elemental abundance

Introduction

The nature of thermoelectric phenomena and materials — competing and contraindicated properties, the complexity and variety of the material systems involved — make it somewhat difficult to develop rational strategies that can lead to significant improvements in performance. Notwithstanding these difficulties, creative approaches have yielded highly promising materials.^{1–8} The guiding principle behind the design of thermoelectric materials, and indeed, any functional material, is to completely understand the causal physics and use such knowledge to rationally optimize material properties. However, even without knowledge of causality, progress across numerous fields is enabled by correlation, without *a priori* understanding of the drivers. Particularly when large datasets are available, robust correlation can be found between seemingly disparate observables. This ability to extract meaningful information from large pools of data has been somewhat under-utilized in the search for new materials.

Thermoelectrics and their historical development have been surveyed by Snyder and Toberer,⁹ Nolas *et al.*,¹⁰ Goldsmid,¹¹ and the two CRC handbooks on thermoelectrics by Rowe.^{12,13} The volume and complexity of research on thermoelectric materials makes the field fertile for a data-driven review — sometimes referred to as data mining, or materials informatics. Informatics-based approaches have been successfully used for estimating some physical properties¹⁴ and the relative stability of selected material systems,¹⁵ and high-throughput methods are becoming increasingly helpful in materials design.¹⁶ Given the volume of information that has been published, knowing the right information to abstract is the first step to such an approach. Furthermore, developing an appropriate visualization strategy to explore the space of thermoelectric materials is crucial. Accordingly, we have reduced the problem of reviewing thermoelectric performance to several key properties at four temperatures of interest, and have created an interactive framework to visualize the large amounts of information. Trends in materials properties emerge from such visualization, and guiding principles for the development of high-performance

thermoelectric materials emerge. Perhaps more importantly, guidelines are suggested for where *not* to look in the parameter space of candidate materials.

Since the efficiency of a thermoelectric device is related to the thermoelectric figure of merit, $zT = S^2T/(\rho\kappa)$, with S the Seebeck coefficient, ρ the electrical resistivity, and κ the thermal conductivity, these different measured parameters were the obvious choice for extraction from publications into a database. A particularly useful format for displaying the large amount of property-based data is based on a modification of the well-known Jonker plot,¹⁷ in which the Seebeck coefficient is usually graphed as a function of the electrical conductivity. The use of the Jonker plot has traditionally been limited to optimizing the carrier concentration in a single thermoelectric material. In contrast, we display *all* materials in a single plot, with the different material classes grouped by marker color. In the modified version employed by us, we prefer to use the electrical resistivity (on a logarithmic scale) ρ as abscissa, and the Seebeck coefficient S as ordinate. The radius of the circular marker represents the performance, which is commonly either zT or the power factor, S^2/ρ .

There is more to a material than just performance. Given the proposed widespread application of thermoelectrics and the potential for high-volume use of materials if certain performance targets are reached, they must be composed of elements that are accessible and not in danger of a supply risk. Crustal abundance of elements, their global production, reserves, and use, are some of the factors that determine supply risk. The criticality of elements in the context of metals that are crucial to energy conversion has been described by Graedel.¹⁸ Himm and Klar¹⁹ have specifically raised these issues for thermoelectric materials. Following the approach of Graedel,¹⁸ we have used our database to calculate several criticality indices for the thermoelectric materials featured here. Resources that are produced almost entirely in a particular region can provide a single entity leverage in determining supplies and prices. This geopolitical influence over materials supply and price can be measured by market concentration, often quantified through the Herfindahl-

1
2
3 Hirschman Index (HHI).^{20,21} The HHI is a financial tool commonly used to measure the
4 monopoly of entities over a commodity or product, and has been previously used as a
5 measure of geopolitical influence on elemental production of a select few elements.^{18,22,23}
6
7 Here we calculate the HHI, based on available 2011 data, of almost all of the first 83
8 elements in the periodic table. Only H, the noble gases, Tc, and Pm are excluded. The HHI
9 is calculated both for elemental production (HHI_P), reflecting the specific geopolitics of
10 the element, as well as for elemental reserves (HHI_R), based on known deposits that could
11 be processed. In conjunction with composition data entered into the database, we employ
12 HHI indices and elemental scarcity values to determine practical issues that influence the
13 likelihood a particular thermoelectric material will find widespread use.
14
15
16
17
18
19
20
21
22
23
24
25

26 Methods

27 Nature and source of data

28
29
30 We have abstracted data from representative publications on a wide variety of thermoelec-
31 tric compounds so that they can be accessed and compared easily with other compounds.
32
33 The data correspond to four different temperatures in the different regimes of interest for
34 high-temperature thermoelectric devices: 300 K, 400 K, 700 K, 1000 K. There are currently
35 over 1100 database entries (rows), each with 17 associated components: temperature,
36 electrical resistivity, Seebeck coefficient, thermal conductivity, power factor, κzT , zT , chem-
37 ical composition, material family, preparatory route, material form (whether single crystal
38 or polycrystalline), author, year of publication, DOI link, and comments. Additionally, we
39 generate new metadata from the chemical composition of materials, such as HHI_P, HHI_R,
40 scarcity, and average atomic weight, \bar{M} . To facilitate comparison of materials, we limit our
41 choice of systems in the database to be bulk, polycrystalline samples. In many instances,
42 we employ the power factor or κzT rather than than the actual figure of merit zT for the
43 visualization, because zT requires κ , the thermal conductivity to be measured. Thermal
44
45
46
47
48
49
50
51
52
53
54
55
56
57
58
59
60

1
2
3 conductivity is not reported for many materials. Furthermore, of all the thermoelectric pa-
4 rameters it is the least reliable and/or reproducible because it is so sensitive to processing
5 conditions.
6
7

8
9 Data from published work was extracted manually from digital publications using free
10 software such as PlotDigitizer²⁴ and DataThief.²⁵ In general, data was extracted from plots
11 of a physical property *vs.* temperature. If data was not explicitly reported at a temperature
12 of interest, values were interpolated, or extrapolated when appropriate. If property traces
13 (curves) were found to rapidly or unpredictably changing in the region of extrapolation,
14 the point was omitted, or data was taken from the nearest reported temperature. In these
15 cases, the temperature of an extrapolated data point is mentioned in the metadata com-
16 ment. Because data was entered by hand, the power factor and figure of merit (zT) were
17 calculated from the extracted data and checked against the reported values to ensure the
18 data was self-consistent.
19
20
21
22
23
24
25
26
27
28
29

30 HHI values based on production and reserves for each element were calculated from
31 2011 USGS commodity statistics following the approach used by others.^{18,22,23} When 2011
32 data was unavailable, data from 2010 or 2009 was used. This has generated a set of HHI
33 values presented in table 1 for much of the periodic table, a significant expansion from
34 previous studies, which focused on eight parent metals.^{18,23} For elements where reserves
35 are seen as adequate or extremely large (*e.g.*, C, O, F, Na, Al, Ca, S, *etc.*) quantitative
36 reserves are not available but their use is unlikely to reduce their availability to critical
37 levels. For elements such as sodium, where multiple commodities are reported separately
38 (carbonates, sulphates, chloride . . .), the reports were combined and the HHI values were
39 generated from the aggregate. Another important consideration is that the production
40 and/or reserve values for a country may be unknown or are withheld. In these instances,
41 where the number of producers and/or the amount of reserves are low, these omissions
42 introduce considerable uncertainty. In these cases, the HHI values were estimated based
43 on the general information provided in the USGS report. Estimates of these elements are
44
45
46
47
48
49
50
51
52
53
54
55
56
57
58
59
60

denoted by an asterisk in table 1.

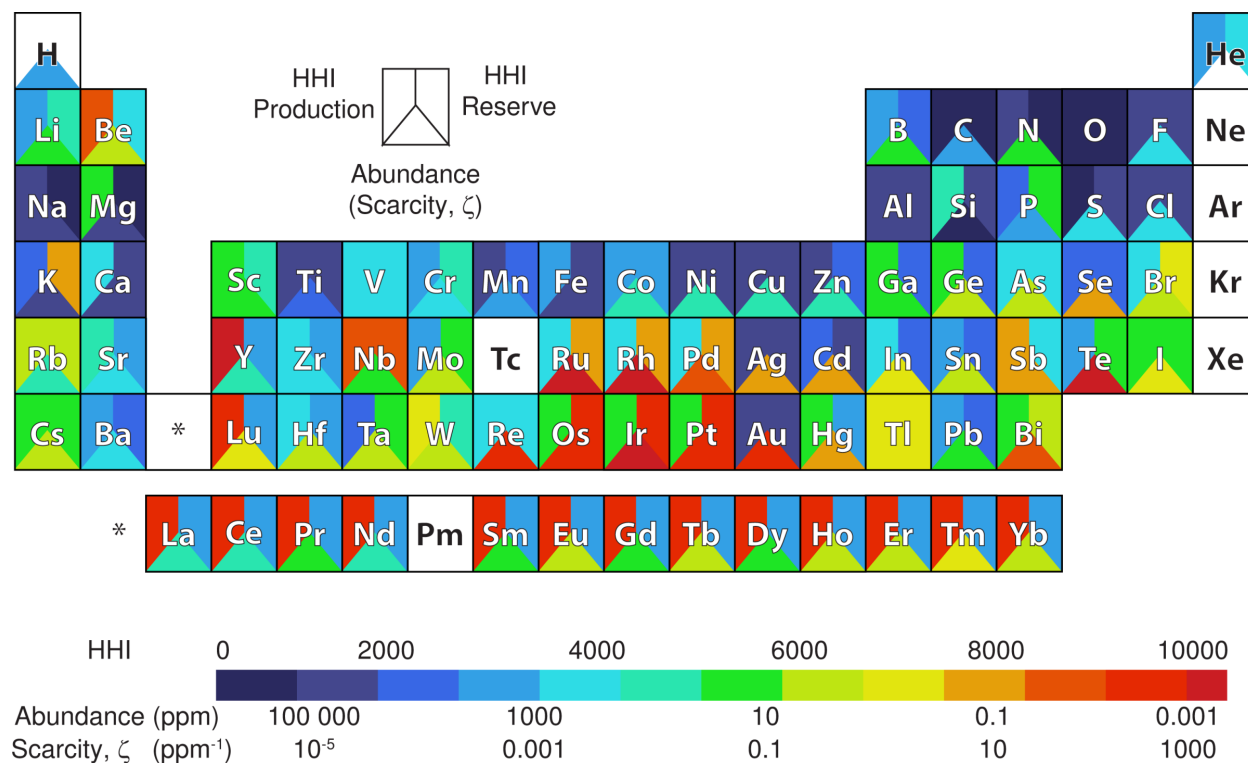


Figure 1: Periodic table indicating elemental scarcity and the HHI (production, reserves) indices for most elements. Values were calculated using USGS statistics from 2011, or if unavailable, from statistics no earlier than 2009.

HHI values were calculated using the following expression, $HHI = \sum_i^N s_i^2$, where s_i is the percent market share of country i in the world production or reserves of a given element. The US Department of Justice and the Federal Trade Commission have designated markets as unconcentrated when $HHI < 1500$, moderately concentrated when the HHI lies between 1500 and 2500, and highly concentrated when $HHI > 2500$.²⁶ If a single country controlled the entire market, $HHI = 100^2$. Elemental HHI values were then used to calculate weighted HHI production and reserve values based on the *weight* fraction of each element in the chemical formula. Atomic weights were taken from the CRC Handbook.²⁷ The crustal abundance of elements were obtained from the CRC Handbook²⁷ and used to generate elemental scarcity values, ζ (crustal abundance in inverse ppm). These scarcity values were used to calculate the effective scarcity of materials based on the weight frac-

Table 1: Herfindahl-Hirschman Index (HHI) calculated for much of the periodic table, using recent USGS data, typically 2011 for most materials, but no older than 2009. Asterisks against numbers indicate some uncertainty (see text).

Element	HHI _P	HHI _R	Element	HHI _P	HHI _R
He	3200	3900	Ru	3200*	8000*
Li	2900	4200	Rh	3200*	8000*
Be	8000	4000*	Pd	3200	8000*
B	2900	2000	Ag	1200	1400
C	500*	500*	Cd	1700	1300
N	1300	500*	In	3300	2000*
O	500*	500*	Sn	2600	1600
F	1500*	1500*	Sb	7900	3400
Na	1100	500*	Te	2900	4900
Mg	5300	500*	I	4900	4800
Al	1600	1000*	Cs	6000*	6000*
Si	4700	1000*	Ba	3000	2300
P	2000	5100	La	9500	3100
S	700	1000*	Ce	9500	3100
Cl	1500*	1500*	Pr	9500	3100
K	1700	7200	Nd	9500	3100
Ca	3900	1500*	Pm	9500	3100
Sc	5500*	4500*	Sm	9500	3100
Ti	1100	1600	Eu	9500	3100
V	3300	3400	Gd	9500	3100
Cr	3100	4100	Tb	9500	3100
Mn	1600	1800	Dy	9500	3100
Fe	2400	1400	Ho	9500	3100
Co	3100	2700	Er	9500	3100
Ni	1000	1500	Tm	9500	3100
Cu	1600	1500	Yb	9500	3100
Zn	1600	1900	Lu	9500	3100
Ga	5500*	1900*	Hf	3400*	2600*
Ge	5300	1900*	Ta	2300	4800
As	3300	4000*	W	7000	4300
Se	2200	1900	Re	3300	3300
Br	3300	6900	Os	5500*	9100*
Rb	6000*	6000*	Ir	5500*	9100*
Sr	4200	3000*	Pt	5500	9100*
Y	9800	2600	Au	1100	1000
Zr	3400	2600	Hg	5500	3100
Nb	8500	8800	Tl	6500*	6500*
Mo	2400	5300	Pb	2700	1800
			Bi	5300	6000

tion of elements in the chemical formula, $\zeta = \sum_i^N (\zeta_i \times \frac{m_i}{m_{tot}})$ where ζ_i and m_i are the scarcity and weight of an element i in a material. The scarcity and Herfindahl-Hirschman Index values (based on production and reserves) for much of the periodic table is shown schematically in figure 1.

Mechanics of visualization

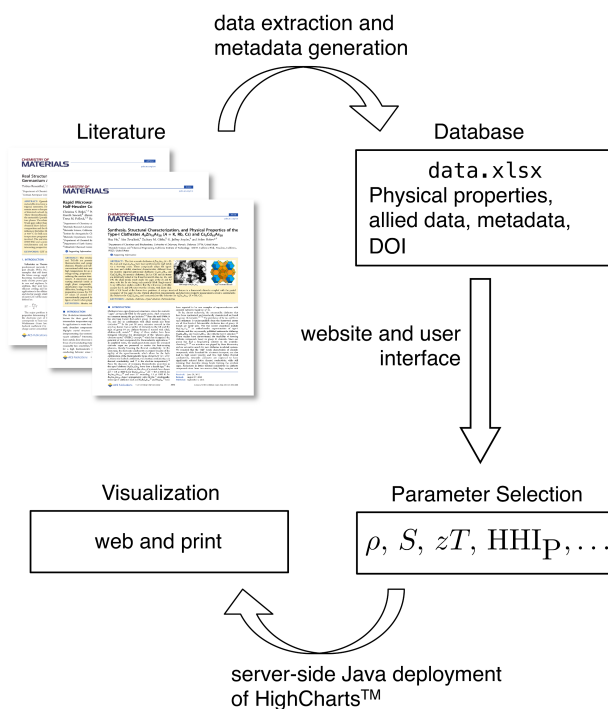
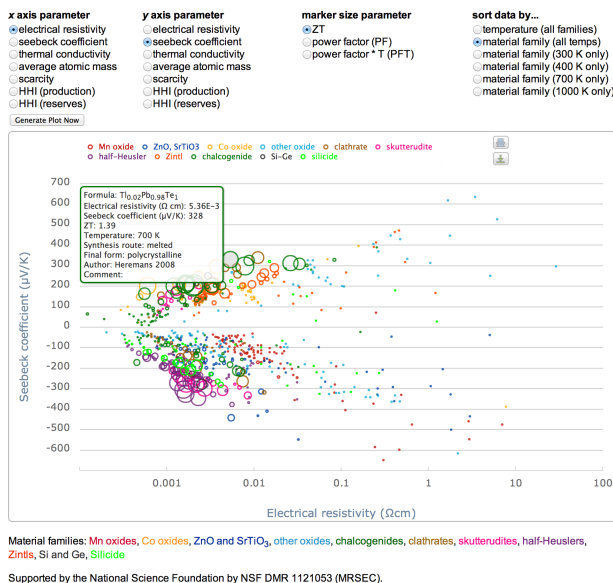


Figure 2: Flowchart for extraction, organization, and visualization of data. Note that the first step of data extraction is manual.

At the initial stages of this work, it became evident that gathering the large amount of data would be futile in the absence of an appropriate framework that permitted the gathered data to be visualized in a flexible manner. In this section we describe a website we have developed and hosted at www.mrl.ucsb.edu:8080/datamine/thermoelectric.jsp as an example of a plausible framework for organizing and visualizing the results of such data mining. The flowchart describing the data mining, database formation, and visualization process is



34
35
36
37
38
39
40
41
42
43
44
45
46
47
48
49
50
51
52
53
54
55
56
57
58
59
60

Figure 3: Screenshot of the web-based visualization tool, that permits the simultaneous visualization of four parameters: abscissa, ordinate, marker size and color: Several variables can be chosen as abscissa and ordinate, and measures of thermoelectric performance can be represented by the radius of the data points. To simplify navigation, families of related materials can be displayed or hidden by clicking their legend marker. Further, hovering over a data point intuitively reveals a tooltip with pertinent information: the names and values for the abscissa and ordinate, the chemical composition, the sample form (e.g., polycrystalline, single crystal, nanoparticles), the preparatory route (e.g., ceramic method, arc melting), the author and year, and either the power factor, κzT , or zT value. Clicking on a data point leads the web-browser directly to the source publication via the document object identifier (DOI).

1
2
3 summarized in figure 2. A screenshot of the website is shown in figure 3. Because several
4 physical parameters were collected and tabulated, visualizing any number of combinations
5 along the abscissa and ordinate is possible using the website. While not all combinations
6 will yield insightful relationships, the website allows using as axes choices, the electrical
7 resistivity, Seebeck coefficient, thermal conductivity, average atomic weight, elemental
8 scarcity, and HHI based on production or reserves of a material. To increase the informa-
9 tion density that can be visualized, a third dimension is also plotted: the size of the data
10 point (the radius of a circle) is proportional to material performance. The marker size
11 can be either power factor (S^2/ρ), κzT , or zT . κzT allows a rough comparison with zT ,
12 but does not require the thermal conductivity to have been measured or reported. Many
13 materials have a thermal conductivity between $1 \text{ W m}^{-1}\text{K}^{-1}$ to $10 \text{ W m}^{-1}\text{K}^{-1}$, and in the
14 case that $\kappa = 1 \text{ W m}^{-1}\text{K}^{-1}$, the numerical value of κzT is the same as the numerical value
15 of zT .
16
17
18
19
20
21
22
23
24
25
26
27
28
29

30 The website allows interactive exploration of all the data. Hovering over a data point
31 reveals a tooltip with pertinent information: the values for the abscissa and ordinate, the
32 chemical composition, the sample form (e.g., polycrystalline, single crystal, nanoparticles),
33 the preparatory route (e.g., ceramic method, arc melting), the author and year, and either
34 the power factor, κzT , or zT value. Additionally, clicking on a data point leads the web-
35 browser directly to the publication *via* the document object identifier (DOI). To enable
36 sorting of the large number of datasets, the user may choose to sort by material family
37 or temperature regime. Although the material families are shown with different marker
38 color the ability to hide or show a given family can make direct comparison more clear.
39 This is accomplished by clicking the name of the dataset in the legend. Finally, option is
40 available to visualize the results of the thermoelectric database employed here, or for users
41 to upload their own data, following an ExcelTM template file available on the website.
42 In this way a user can look for trends in their data or use the website's JavaTM code to
43 generate additional non-performance-related data such as scarcity, average atomic weight
44
45
46
47
48
49
50
51
52
53
54
55
56
57
58
59
60

or HHI.

Results and discussion

Examples of useful visualization schemes

Table 2: The flexibility of the visualization framework allows users to examine the relation between any number of different parameters. Several examples are tabulated here, along with what can potentially be learned.

Abscissa	Ordinate	Size (radius)	Use	Finding
ρ	S	$S^2/\rho, \kappa zT, zT$	General trends, insight	Compound must be metallic for high performance
\bar{M}	κ	$\kappa zT, zT$	Effect of $\bar{\omega}$	κ decreases with increasing \bar{M}
$\rho_{300\text{ K}}$	$S_{300\text{ K}}$	$\kappa zT, zT$ at different temperatures	estimate high- T performance	Rapid screening of high- T materials
ρ	κ	$\kappa zT, zT$	Effect of κ_e	Slope of κ vs. ρ changes linearly with T of dataset
HHI _P	HHI _R	$\kappa zT, zT$	Material choice and criticalilty	Intensive use of rare-earths and/or Sb may strain markets
ζ	HHI _P	$\kappa zT, zT$	Material choice and criticalilty	Some elements are abundant, yet will have volatile prices. Many state-of-the-art high-performance materials rank poorly in criticality indices

In table 2, we list those combinations of parameters that we have found particularly useful to plot. The various combinations and the findings are described more thoroughly in the following sections.

Rapid screening of materials

We begin by visualizing the general trend in Seebeck coefficient as a function of the electrical resistivity (figure 4): These two properties jointly contribute to zT . Higher information density is achieved by encoding the power factor as the marker size. This style of plot

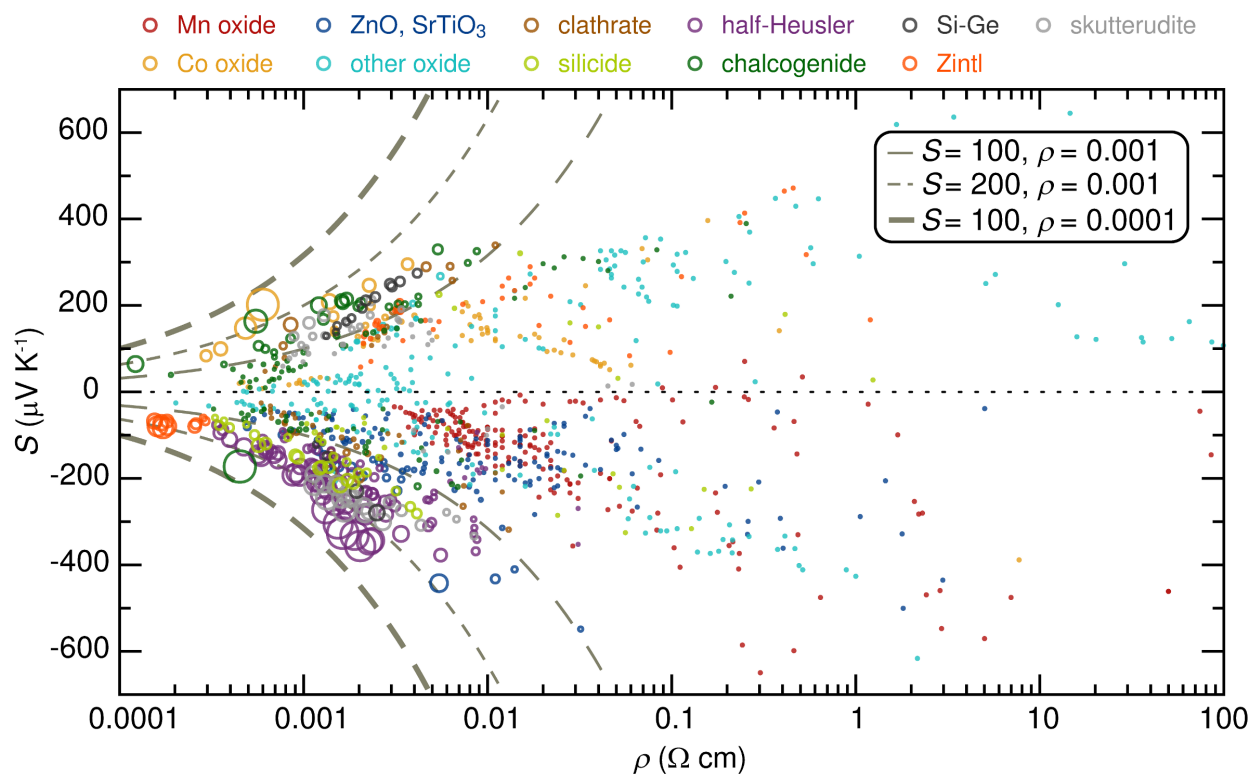


Figure 4: The Seebeck coefficient of a wide variety of materials grouped by material family is plotted against electrical resistivity. The marker size (radius) here is proportional to the power factor. Materials with high performance exhibit an high Seebeck coefficient at a given electrical resistivity, and lie outside the conical envelope defined by most thermoelectric materials. This envelope can be described by lines of constant power factor (S^2/ρ), which are shown to give lower and upper bounds of performance. With few exceptions, all investigated materials with reasonable thermoelectric performance are well described as metals ($\rho_{300\text{K}} \ll 0.01 \text{ } \Omega \text{ cm}$).

1
2
3 was first introduced in a previous report.²⁸ Here we enhance it by assigning a color to
4 a family of related materials. Although the axes are similar to those for a Jonker plot
5 (where Seebeck coefficient is plotted vs. electrical conductivity) the application is quite
6 distinct. Jonker plots are usually employed to examine the effect of changing the carrier
7 concentration in a single material.^{17,29}

8
9
10
11
12
13 Looking at the data as a whole, general envelope of materials is cone-like, widening
14 at higher electrical resistivities (figure 4). The best materials clearly define the left-most
15 edges of the envelope. All the high-performance materials have interesting physics that
16 make them special, even when looking only at the these two properties. Specifically, mate-
17 rials with metallic behavior generally have low Seebeck coefficients, but high-performance
18 thermoelectric materials violate this principle. For example, band asymmetry or high
19 band degeneracy near the Fermi level in BiTe- and PbTe-based systems leads to an un-
20 usually high Seebeck coefficient, despite their metallic behavior.^{5,30} Likewise, even though
21 Na_xCoO₂ is metallic, with polycrystalline samples having $\rho_{300\text{K}} \approx 0.002 \Omega \text{ cm}$, the corre-
22 lated behavior of electrons and spin contribution to thermopower—or, arguably, the unique
23 band structure—lead to a remarkably high Seebeck coefficient.^{31,32}

24
25
26
27
28
29
30
31
32
33
34
35
36 Looking at the ensemble of data (figure 4) also provides insight about particular fami-
37 lies of materials. For example, we find that optimizing non-metallic moderate-performance
38 materials is unlikely to lead to dramatic gains in performance, as the electrical resistivity is
39 too high. For example, CaMnO₃-based systems remain a topic of intense study, but when
40 viewed in the context of all materials, their placement in the map suggests it is unlikely
41 that high performance will ever be reached.

42
43
44
45
46
47
48
49
50
51
52
53
54
55
56
57
58
59
60
It was previously shown that room-temperature thermoelectric properties (Seebeck
coefficient and electrical resistivity) of several high-performance thermoelectric materi-
als clustered in one area of the thermoelectric map (figure 4) when compared with low-
performance materials.²⁸ Here we suggest that the relationship between the properties at
room-temperature and the properties at the temperature of highest zT for a small set of

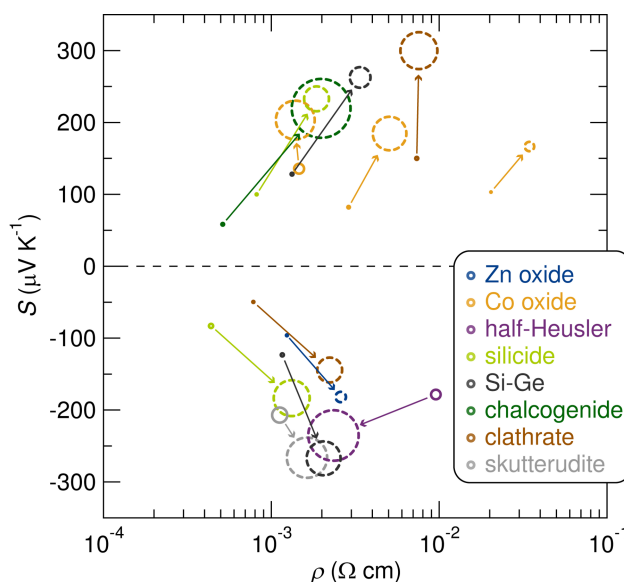


Figure 5: Several high-performance thermoelectric materials were chosen to examine how their room temperature properties relate to their properties at the temperature of highest zT . Arrows show the change from room temperature (solid circles) to the temperature of highest zT (dashed circles). The properties of several different material families appear to cluster at their temperature of highest zT , with the electrical resistivity between $0.001 \Omega \text{ cm}$ and $0.01 \Omega \text{ cm}$ and the absolute value of the Seebeck coefficient between $150 \mu\text{V K}^{-1}$ and $300 \mu\text{V K}^{-1}$. Additionally, nearly all high-performance materials have a room-temperature electrical resistivity below the Mott maximum metallic resistivity ($\rho_{300\text{K}} < 0.01 \Omega \text{ cm}$). The materials shown are $\text{Zn}_{0.98}\text{Al}_{0.02}\text{O}$,³³ Na_xCoO_2 ,³⁴ $\text{Ca}_2\text{Co}_2\text{O}_5$,³⁵ $(\text{Zr}_{0.5}\text{Hf}_{0.5})_{0.5}\text{Ti}_{0.5}\text{NiSn}$,³⁶ $\text{Mg}_2\text{Si}_{0.999}\text{Bi}_{0.001}$ (*p*-type),³⁷ $(\text{Mg}_2\text{Si})_{0.97}\text{Bi}_{0.03}$ (*n*-type),³⁸ $\text{Si}_{0.8}\text{Ge}_{0.2}$ (*p*-type),³⁹ $\text{Si}_{0.8}\text{Ge}_{0.2}$ (*n*-type),³⁹ $\text{PbTe}_{0.75}\text{Se}_{0.25}$,⁵ CsBi_4Te_6 ,⁴⁰ $\text{Ba}_8\text{Ga}_{18}\text{Ge}_{28}$ (*p*-type),⁴¹ $\text{Ba}_8\text{Ga}_{16}\text{Ge}_{30}$ (*n*-type),⁴² and $\text{In}_{0.25}\text{Co}_4\text{Sb}_{12}$.⁴³

1
2
3 mid- and high-temperature thermoelectric materials (figure 5). Interestingly, the proper-
4 ties of several different material families appear to cluster at their temperature of high-
5 est zT , with the electrical resistivity between $0.001 \Omega \text{ cm}$ and $0.01 \Omega \text{ cm}$ and the absolute
6 value of the Seebeck coefficient between $150 \mu\text{VK}^{-1}$ and $300 \mu\text{VK}^{-1}$. Although the small
7 dataset prevents any strong conclusions, examining more materials may provide some
8 predictive ability of high-temperature properties, using those measured at room temper-
9 ature. Even some hints with regard to high-temperature properties would be powerful:
10 measurement of high temperature properties is time-consuming and requires specialized
11 instruments. Furthermore, there is much more room-temperature data available in the lit-
12 erature and a single room-temperature measurement could facilitate combinatorial testing
13 of a large phase space.⁴⁴

14
15
16
17
18
19
20
21
22
23
24
25
26 The thermoelectric survey shown in figure 4 and the high-temperature trends observed
27 in figure 5 suggest that all thermoelectric materials with any appreciable performance are
28 all metallic, with an electrical resistivity well below the Mott maximum metallic resistiv-
29 ity at room temperature, *i.e.* $\rho_{300\text{K}} < 0.01 \Omega \text{ cm}$. This provides a valuable thermoelectric
30 screening criterion and guiding direction for future studies. Unlike intermetallic com-
31 pounds, which are generally metallic or semi-metallic, transition metal oxides span the
32 gamut of electrical resistivities, ranging from insulating (*e.g.* TiO_2) to metallic (*e.g.* ReO_3).
33 A common strategy to seeking effective thermoelectric oxides is to examine the metal/non-
34 metal border, *i.e.* proximal to the Mott minimum metallic conductivity, where Seebeck
35 coefficients are often substantial. However, the utility of this approach appears somewhat
36 questionable, given that the good thermoelectrics appear well on the metallic side.

50 Thermal conductivity

51
52
53 Over the past several decades, the majority of improvements in thermoelectric materials
54 have resulted from decreasing the lattice contribution to thermal conductivity. A num-
55 ber of materials selection guidelines have been identified in order to reduce the thermal
56
57
58
59
60

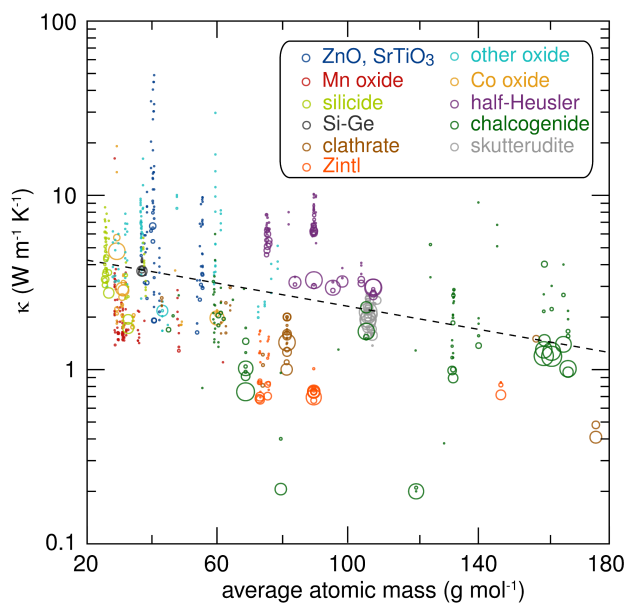


Figure 6: Thermal conductivity plotted against average atomic weight for a variety of thermoelectric materials grouped by material family, with marker indicating zT . The dashed line represents a best fit regression. Notwithstanding the combination of pure phase materials and heavily nano-structured materials, the thermal conductivity generally decreases with increasing average atomic weight. However, at fixed average atomic weight, there is considerable tunability.

conductivity in a material.^{9,45,46} One such strategy is to lower the vibrational frequency, and thus the thermal conductivity, by using materials with a large average atomic weight. This has frequently been touted as a reason for the higher thermal conductivity in oxides relative to compounds with heavier anions such as Te, Se, and Sb. Examination of the relationship between thermal conductivity, κ , and average atomic weight, \bar{M} , confirms the general reduction of thermal conductivity in heavier compounds (figure 6). For example, compounds with $\bar{M} \approx 25$ g/mol have an average $\kappa \approx 4$ W m⁻¹K⁻¹ whereas those with $\bar{M} \approx 105$ g/mol have an average $\kappa \approx 2$ W m⁻¹K⁻¹. In the absence of point defects or other scattering mechanisms, the reduction in thermal conductivity should scale as $\bar{M}^{-1/2}$.⁴⁷ However, many of the materials contained in this plot rely on additional techniques to reduce thermal conductivity (*e.g.*, increased phonon scattering from defects, alloying, grain boundaries, interfaces, nano-bulk compounds, complex crystal structures, *etc.*). On the other hand, the lower vibrational frequency achieved in compounds with large average

atomic weight has only a small impact on the electrical resistivity.

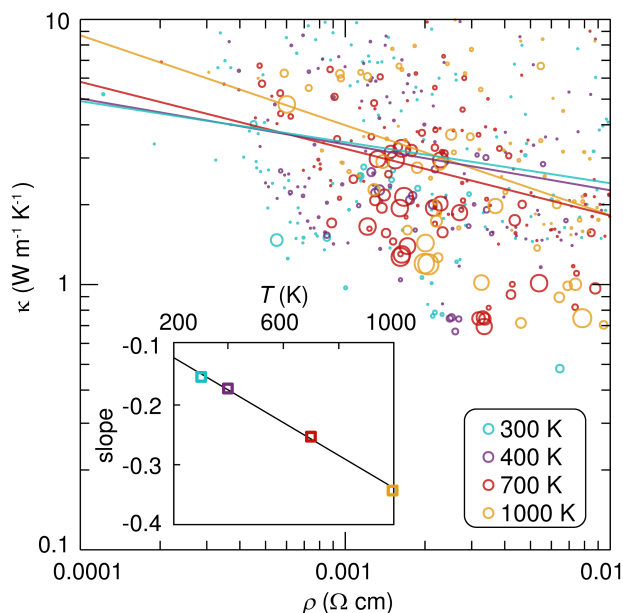


Figure 7: Thermal conductivity plotted against electrical resistivity for a variety of thermoelectric materials grouped by dataset temperature with marker size indicating zT . Lines are best fit linear regressions of the data plotted on log-log plot. The slopes are then plotted as a function of dataset temperature in the inset. Despite the large scatter in the data that has been fitted, the general trend that emerges is consistent with the T^{-1} dependence of Umklapp scattering of phonons.

Finally, we examine the relationship between total thermal conductivity and electrical resistivity (figure 7); the thermal conductivity decreases with increasing electrical resistivity. This behavior is expected for the electronic contribution from the Wiedemann-Franz law, $\kappa_e/\sigma = \pi^2 k_B^2/3e^2 T = L_0 T$ where L_0 is the Lorenz number ($2.44 \times 10^{-8} \text{ W } \Omega \text{ K}^{-2}$) and σ is the electrical conductivity. There is another consideration that could be important. For a given composition, the same structural features that give rise to low resistivity, — for example structures that are highly connected in three-dimensions — are the same features that often lead to high lattice thermal conductivity.

Examining the broad correlation across all material families in the metallic region ($\rho < 0.01 \text{ } \Omega \text{ cm}$) reveals the slope of the log-log plot κ vs. ρ decreases linearly with increasing temperature (inset, figure 7). The power-law behavior revealed from the fit is consistent with Umklapp scattering, explained below.

Examining correlations across many material families with disparate properties will only provide general trends rather than precise values for a specific material family. Nevertheless, there is insight in comparing the correlation (*i.e.* slope) at different temperatures. For example, while there is considerable scatter in the data at each temperature, the temperature dependence of the correlation is strongly linear ($R^2 = 0.997$), in good agreement with the expected behavior as Umklapp scattering begins to dominate at higher temperatures. The changing slope as a function of dataset temperature results from increased phonon scattering lowering the phonon contribution to thermal conductivity at higher temperatures. When the lattice κ_l is minimized, the electronic κ_e dominates, so changes in electrical resistivity have a larger impact on thermal conductivity. The power dependence of temperature on κ vs. ρ slope may result from the Umklapp scattering, a three-phonon scattering process. According to the Bose-Einstein expression, $\langle n \rangle = 1/(\exp[\hbar\omega/k_B T] - 1) \approx k_B T/\hbar\omega$, the average phonon population, $\langle n \rangle$, increases with temperature which increases the likelihood of phonon-phonon scattering. In fact, Grimvall⁴⁷ has predicted a T^{-1} temperature dependence for three-phonon scattering at moderate temperatures (near the Debye temperature).

Resource considerations

In this section we review parameters beyond thermoelectric performance. We analyze resource considerations such as scarcity and supply risk of thermoelectric materials based on their elemental composition. The implications of the analysis on the choice of thermoelectric materials is discussed below. A combined analysis of performance and resource considerations is necessary for identifying thermoelectric materials with the greatest promise of widespread application. Scarce elements could be employed in efforts to make high-impact discoveries and to obtain better understanding of materials trends. However, when materials are meant for widespread deployment, the incorporation of scarce elements becomes

an important point to consider during the design phase.

We begin by discussing scarcity which places a fundamental limit on the amount of material available for use. The following section addresses HHI, a measure of geopolitical influence. Abundance of raw materials is a key factor that determines cost, in addition to other factors such as actuarial costs, packaging, transportation, and assembly, to name a few. Technologies relying on scarce elements such as Re, Te, or Pt group metals, are susceptible to large materials costs. For example, turbine engines consume 70% of the world's Re production, which drives the cost of Re.⁴⁸ The cost of scarce materials can be exacerbated when multiple technologies use the same element, such as the use of In for transparent conducting oxides (TCO) and copper indium gallium selenide (CIGS) photo-voltaics. Cost can also be high for relatively abundant materials where both the demand and production are low, such as low purity Si.¹⁹

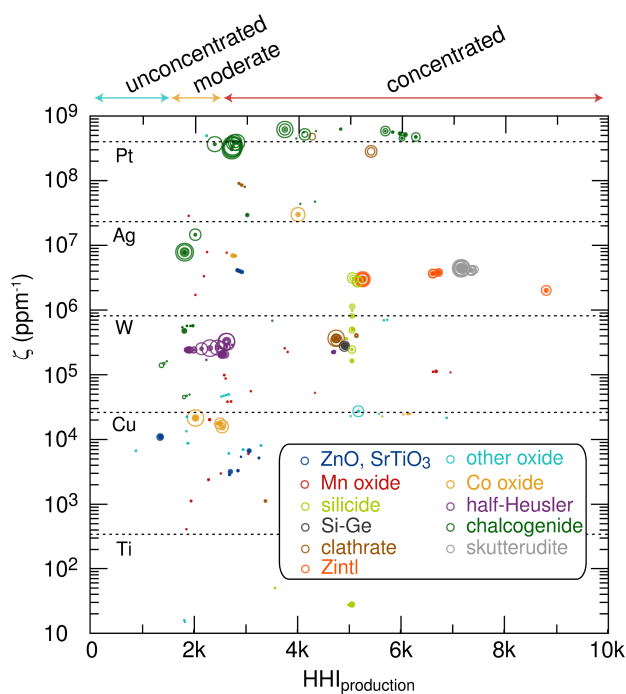


Figure 8: Nearly all high-performance thermoelectric materials pose either a scarcity or HHI risk, which will limit any intensive deployment for energy applications. The scarcity (inverse crustal abundance) of familiar elements is shown to give context. Although not immediately apparent, incorporation of Bi, Sb, Te, or rare-earth elements leads to significant risk in either scarcity or HHI. Materials that pose a lower risk of criticality are SiGe, silicides, half-Heuslers, and cobaltates.

1
2
3
4
5
6
7
8
9
10
11
12
13
14
15
16
17
18
19
20
21
22
23
24
25
26
27
28
29
30
31
32
33
34
35
36
37
38
39
40
41
42
43
44
45
46
47
48
49
50
51
52
53
54
55
56
57
58
59
60

Plotting scarcity against HHI_p for thermoelectric materials is a useful tool to identify technologies where materials cost can be a concern. About half of the highest efficiency materials contain a large weight fraction of Te. (e.g. Be_2Te_3 , PbTe , La_3Te_4 , TAGS, Ag_9TlTe_5 , SbTe , or Tl_9BiTe_6 .) Given that Te has one of the highest scarcity values, these materials may pose problems for widespread deployment. Te may be rare, but its HHI_p is not critical because it is produced in many countries as a byproduct of Cu, reducing supply risks. Additionally, with proper recycling and waste processing policies, Te could be recovered and reprocessed, a practice already carried out by some photovoltaic companies.

An unanticipated finding from analysis of figure 8 was that incorporation of even small amounts of scarce elements in otherwise Earth-abundant compounds may be problematic. Mg_2Si has a low scarcity of $\zeta = 27.3$, but when doped with trace amounts of Bi to $\text{Mg}_2\text{Si}_{0.993}\text{Bi}_{0.007}$,³⁷ ζ increases to 1.1×10^6 because of the extremely low crustal abundance (0.025 ppm) and high atomic weight of Bi. Even 0.1% substitution of Bi on the Si site results in a four orders of magnitude increase in ζ .

A compromise between performance and resource considerations is seen in oxide materials. Cobalt oxides, such as NaCo_2O_4 , $\text{Ca}_2\text{Co}_2\text{O}_5$, and $\text{Ca}_3\text{Co}_4\text{O}_9$, as well as the reduced ferroelectric oxide $\text{Sr}_{0.61}\text{Ba}_{0.39}\text{Nb}_2\text{O}_6$, have competitive zT values with scarcity values four orders of magnitude lower than state-of-the-art chalcogenides, two orders of magnitude lower than skutterudites ($\text{ACo}_4\text{Sb}_{12}$, A = caged atom) and Zintlites (Zn_4Sb_3 , $\text{Yb}_{14}\text{MnSb}_{11}$), and one order of magnitude lower than clathrates ($\text{Ba}_8\text{Ga}_{18}\text{Ge}_{28}$) and half-Heuslers (TiNiSn). ZnO and SrTiO_3 are even more abundant, but have lower performance. It is worth noting that, as with Mg_2Si , if Bi is added to improve performance (e.g., $\text{Bi}_2\text{Sr}_2\text{Co}_2\text{O}_8$), a dramatic increase in ζ is observed.

Determining the Herfindahl-Hirschman Index (HHI) for all elements has shown the following elements to be most at risk for market abuse: Be, K, Br, Rb, Nb, Sb, Cs, W, the rare-earth (RE) elements including Y, and the platinum group elements (PGE): Ru, Rh, Pd, Os, Ir, Pt. These elements have HHI values of greater than 6000 in either production or

1
2
3 reserves. As with scarcity, introduction of even small amounts of elements with high HHI_P
4 or HHI_R such as REs, Sb, or Tl, may not be justified for a marginal gain in performance.
5
6 Materials like Zintl or clathrate antimonides and rare-earth containing chalcogenides are
7
8 poor outliers compared to other thermoelectric materials in terms of HHI_P risk.
9
10

11 The importance of HHI when considering thermoelectric materials can be illustrated
12 through recent price spikes in antimony. In 2011, a single country that was responsible for
13 nearly 90% of Sb mining halted much of its production. As a result, Sb prices rose over
14 15% in less than a year. While Sb is particularly concentrated, over 70% of elements ex-
15 ceed the 2500 threshold to be considered “highly concentrated” according to Federal Trade
16 Commission definition. This makes many thermoelectric materials sensitive to changing
17 market conditions. This analysis signals that emphasis should be placed on developing ma-
18 terials with a lower HHI. Examples of materials that have reasonable performance and low
19 HHI include S- and Se-containing chalcogenides, the half-Heusler materials low in Hf- and
20 Nb-content, and sodium cobalt oxides. Practical consideration of these criticality param-
21 eters, although not performance related, would benefit both material and experimental
22 design.
23
24
25
26
27
28
29
30
31
32
33
34
35
36
37
38

39 **Future directions**

40
41
42 Thermoelectric data mining provides insights, enabling rational material design. There
43 are, in addition to the properties described, a number of additional materials properties
44 and parameters that were not included that are nevertheless highly desired in comparing
45 different thermoelectric materials. For example, crystallographic data would allow for the
46 calculation of average atomic volume, unit cell volume, and number of atoms in the unit
47 cell. This could be valuable in looking at changes in thermal conductivity. Mechanical
48 properties, coefficients of thermal expansion, and high temperature stability indices would
49 be valuable in thermoelectric device design. Furthermore, additional non-performance re-
50
51
52
53
54
55
56
57
58
59
60

lated parameters such as toxicity are of great interest in environmental and life-cycle analyses of thermoelectric components. Some of these properties could be extracted from other databases, measured, or approximated. Other properties, such as toxicity, are fundamentally more challenging to obtain. A comprehensive approach to comparing toxicity across compounds is not well established, since it depends on environmental transformations and persistence, exposure route, form (e.g., oxidation state, presence of counter-ions), and many other factors. Finally, adding a dimension of time to the plots we have shown may increase their utility. For example, looking at the evolution of HHI over time may provide a better ability to estimate risk and market volatility of particular materials.

The data-driven approach and visualization can be readily extended to other materials functions. For example, photovoltaics, thermal barrier coatings, dielectrics, fuel cells, transparent conducting oxides, gas turbine superalloys, and batteries are all areas of research where progress depends on optimizing several competing requirements concurrently, that could benefit from a comprehensive, data-driven approach.

Closing remarks

The framework demonstrated here allows researchers obtain a birds-eye view of a large domain of thermoelectrics research, and compare new materials of interest. The thermoelectric design space is large, and this overview of the field, including important non-performance related parameters, allows researchers to focus on property regions and material families best suited for a given application. In the case of power generation with thermoelectrics, analysis of HHI, scarcity, and materials properties can be done visually. Using the methods shown here, we highlight several families of materials that combine good performance-related properties with non-critical resource availability: highly conductive (electrically) early transition metal oxides, silicides with low bismuth content, and half-Heuslers. We also observe that all high- zT materials are found in the metallic con-

1
2
3
4
5
6
7
8
9
10
11
12
13
14
15
16
17
18
19
20
21
22
23
24
25
26
27
28
29
30
31
32
33
34
35
36
37
38
39
40
41
42
43
44
45
46
47
48
49
50
51
52
53
54
55
56
57
58
59
60

duction region 0.001 Ω cm to 0.01 Ω cm. This visualization framework is also valuable from a design perspective, where it serves as a useful guide to identifying promising new materials.

Acknowledgement

We thank the National Science Foundation for support of this research through NSF-DMR 1121053. RS acknowledges useful discussions with Ambuj Singh. MWG thanks the Natural Sciences and Engineering Council of Canada for support through a NSERC Postgraduate Scholarship and the US Department of State for an International Fulbright Science & Technology Award. The research at Harvard was supported by the Office of Naval Research through contract N000141110894 and by the Harvard University Center for the Environment.

Appendix

Table 3: General class of material family and publications from which data was extracted. Representative publications are chosen for each class of materials.

Material family	References
Mn oxide	49–60
ZnO, SrTiO ₃	33,61–72
Co oxide	34,35,73–77
other oxide	28,63,78–96
Si-Ge	39,97
clathrate	41,42,98–103
half-Heusler	36,104–107
skutterudite	1,7,43,108
chalcogenide	5,30,40,109–122
silicide	37,38,123–126
Zintl	127–132

References

- [1] Sales, B. C.; Mandrus, D.; Williams, R. K. *Science* **1996**, *272*, 1325–1328.
- [2] Hicks, L. D.; Dresselhaus, M. S. *Phys. Rev. B* **1993**, *47*, 12727–12731.
- [3] Dresselhaus, M. S.; Chen, G.; Tang, M. Y.; Yang, R. G.; Lee, H.; Wang, D. Z.; Ren, Z. F.; Fleurial, J. P.; Gogna, P. *Adv. Mater.* **2007**, *19*, 1043–1053.
- [4] Kauzlarich, S. M.; Brown, S. R.; Snyder, G. J. *Dalton Trans.* **2007**, 2099–2107.
- [5] Pei, Y.; Shi, X.; LaLonde, A.; Wang, H.; Chen, L.; Snyder, G. J. *Nature* **2011**, *473*, 66–69.
- [6] Hsu, K. F.; Loo, S.; Guo, F.; Chen, W.; Dyck, J. S.; Uher, C.; Hogan, T.; Polychroniadis, E. K.; Kanatzidis, M. G. *Science* **2004**, *303*, 818–821.
- [7] Biswas, K.; Good, M. S.; Roberts, K. C.; Subramanian, M. A.; Hendricks, T. J. *J. Mater. Res.* **2011**, *26*, 1827–1835.
- [8] Biswas, K.; He, J.; Blum, I. D.; Wu, C.-I.; Hogan, T. P.; Seidman, D. N.; Dravid, V. P.; Kanatzidis, M. G. *Nature* **2012**, *489*, 414–418.
- [9] Snyder, G. J.; Toberer, E. S. *Nat. Mater.* **2008**, *7*, 105–114.
- [10] Nolas, G. S.; Sharp, J.; Goldsmid, H. J. *Thermoelectrics : Basic principles and new materials developments*; Springer series in materials science 45; Springer: Berlin ; New York, 2001; p 292.
- [11] Goldsmid, H. J. *Introduction to thermoelectricity*; Springer: Heidelberg, 2009.
- [12] Rowe, D. M. *CRC Handbook of thermoelectrics*; CRC Press: Boca Raton, FL, 1995.
- [13] Rowe, D. M. *Thermoelectrics handbook: Macro to nano*; CRC press: Boca Raton, FL, 2006.

- 1
2
3 [14] Bucholz, E. W.; Kong, C. S.; Marchman, K. R.; Sawyer, W. G.; Phillpot, S. R.; Sin-
4 nott, S. B.; Rajan, K. *Tribol. Lett.* **2012**, *47*, 211–221.
5
6
7
8 [15] Kong, C. S.; Villars, P.; Iwata, S.; Rajan, K. *Comput. Sci. Discovery* **2012**, *5*, 015004.
9
10
11 [16] Curtarolo, S.; Hart, G. L. W.; Nardelli, M. B.; Mingo, N.; Sanvito, S.; Levy, O. *Nature*
12 *Materials* **2013**, *12*, 191–201.
13
14
15
16 [17] Jonker, G. H. *Philips Res. Rep.* **1968**, *23*, 131–138.
17
18
19 [18] Graedel, T. E. *Annu. Rev. Mater. Res.* **2011**, *41*, 323–335.
20
21
22 [19] Homm, G.; Klar, P. J. *Phys. Status Solidi RRL* **2011**, *5*, 324–331.
23
24
25 [20] Herfindahl, O. C. Concentration in the US Steel Industry. Ph.D. thesis, Columbia
26 University, 1950.
27
28
29 [21] Hirschman, A. O. *National power and the structure of foreign trade*; California Uni-
30 versity Press: Berkeley, 1980.
31
32
33
34 [22] Rosenau-Tornow, D.; Buchholz, P.; Riemann, A.; Wagner, M. *Resour. Policy* **2009**,
35 *34*, 161–175.
36
37
38
39 [23] *Fact Sheet: Rare Earth Oxides (REO)*; POLINARES Working Paper, 2012; Vol. 37.
40
41
42 [24] Huwaldt, J. A. Plot Digitizer. 2011; <http://plotdigitizer.sourceforge.net/>.
43
44
45
46
47 [25] Tummers, B. DataThief III. 2006; <http://datathief.org/>.
48
49
50 [26] U.S. Department of Justice and the Federal Trade Commission, *Horizontal merger*
51 *guidelines*; 2010.
52
53
54
55 [27] Haynes, W., Ed. *CRC Handbook of chemistry and physics*, 93rd ed.; CRC Press: Boca
56 Raton, FL, 2012.
57
58
59
60

- 1
2
3 [28] Gaultois, M. W.; Barton, P. T.; Birkel, C. S.; Misch, L. M.; Rodriguez, E. E.;
4 Stucky, G. D.; Seshadri, R. *J. Phys.: Condens. Matter* **2013**, *25*, 186004.
5
6
7
8 [29] Bak, T.; Nowotny, J. *J. Phys. Chem. C* **2011**, *115*, 9746–9752.
9
10
11 [30] Heremans, J. P.; Jovovic, V.; Toberer, E. S.; Saramat, A.; Kurosaki, K.; Charoen-
12 phakdee, A.; Yamanaka, S.; Snyder, G. J. *Science* **2008**, *321*, 554–557.
13
14
15 [31] Koshibae, W.; Tsutsui, K.; Maekawa, S. *Phys. Rev. B* **2000**, *62*, 6869–6872.
16
17
18 [32] Singh, D. J.; Kasinathan, D. *J. Electron. Mater.* **2007**, *36*, 736–739.
19
20
21 [33] Tsubota, T.; Ohtaki, M.; Eguchi, K.; Arai, H. *J. Mater. Chem.* **1997**, *7*, 85–90.
22
23
24 [34] Fujita, K.; Mochida, T.; Nakamura, K. *Jpn. J. Appl. Phys., Part 1* **2001**, *40*, 4644–
25 4647.
26
27
28 [35] Funahashi, R.; Matsubara, I.; Ikuta, H.; Takeuchi, T.; Mizutani, U.; Sodeoka, S. *Jpn.*
29 *J. Appl. Phys., Part 2* **2000**, *39*, L1127–L1129.
30
31
32 [36] Sakurada, S.; Shutoh, N. *Appl. Phys. Lett.* **2005**, *86*, 082105.
33
34
35 [37] Bux, S. K.; Yeung, M. T.; Toberer, E. S.; Snyder, G. J.; Kaner, R. B.; Fleurial, J. P. *J.*
36 *Mater. Chem.* **2011**, *21*, 12259–12266.
37
38 [38] Fukano, M.; Iida, T.; Makino, K.; Akasaka, M.; Oguni, Y.; Takanashi, Y. *MRS Proc.*
39 **2007**, *1044*, 1044–U06–13.
40
41
42 [39] Vining, C. B.; Laskow, W.; Hanson, J. O.; Vanderbeck, R. R.; Gorsuch, P. D. *J. Appl.*
43 *Phys.* **1991**, *69*, 4333–4340.
44
45
46 [40] Chung, D. Y.; Hogan, T.; Brazis, P.; Rocci-Lane, M.; Kannewurf, C.; Bastea, M.;
47 Uher, C.; Kanatzidis, M. G. *Science* **2000**, *287*, 1024–1027.
48
49
50
51
52
53
54
55
56
57
58
59
60

- 1
2
3 [41] Anno, H.; Hokazono, M.; Kawamura, M.; Nagao, J.; Matsubara, K. *21st International Conference on Thermoelectrics*; 2002; pp 77–80.
4
5
6
7
8 [42] Toberer, E. S.; Christensen, M.; Iversen, B. B.; Snyder, G. J. *Phys. Rev. B* **2008**, *77*,
9 075203.
10
11
12 [43] He, T.; Chen, J. Z.; Rosenfeld, H. D.; Subramanian, M. A. *Chem. Mater.* **2006**, *18*,
13 759–762.
14
15
16
17 [44] Funahashi, R.; Mikami, M.; Urata, S.; Kitawaki, M.; Kouuchi, T.; Mizuno, K. *Meas.*
18 *Sci. Technol.* **2005**, *16*, 70–80.
19
20
21
22 [45] DiSalvo, F. J. *Science* **1999**, *285*, 703–706.
23
24
25 [46] Clarke, D. R. *Surf. Coat. Technol.* **2003**, *163*, 67–74.
26
27
28 [47] Grimvall, G. *Thermophysical properties of materials*, 1st ed.; Elsevier: Amsterdam,
29 1999.
30
31
32 [48] Naumov, A. *Russ. J. Non-Ferrous Met.* **2007**, *48*, 418–423.
33
34
35 [49] Lan, J.; Lin, Y.; Mei, A.; Nan, C.; Liu, Y.; Zhang, B.; Li, J. *J. Mater. Sci. Technol.* **2009**,
36 *25*, 535–538.
37
38
39 [50] Park, J. W.; Kwak, D. H.; Yoon, S. H.; Choi, S. C. *J. Alloys Compd.* **2009**, *487*,
40 550–555.
41
42
43 [51] Populoh, S.; Trottmann, M.; Aguire, M. H.; Weidenkaff, A. *J. Mater. Res.* **2011**, *26*,
44 1947–1952.
45
46
47 [52] Zhou, Y. Q.; Matsubara, I.; Funahashi, R.; Xu, G. J.; Shikano, M. *Mater. Res. Bull.*
48 **2003**, *38*, 341–346.
49
50
51 [53] Lan, J. L.; Lin, Y. H.; Fang, H.; Mei, A.; Nan, C. W.; Liu, Y.; Xu, S. L.; Peters, M. J.
52 *Am. Ceram. Soc.* **2010**, *93*, 2121–2124.
53
54
55
56
57
58
59
60

- 1
2
3
4 [54] Flahaut, D.; Mihara, T.; Funahashi, R.; Nabeshima, N.; Lee, K.; Ohta, H.;
5 Koumoto, K. *J. Appl. Phys.* **2006**, *100*, 084911.
6
7
8 [55] Kobayashi, T.; Takizawa, H.; Endo, T.; Sato, T.; Shimada, M.; Taguchi, H.; Na-
9 gao, M. *J. Solid State Chem.* **1991**, *92*, 116–129.
10
11
12 [56] Sparks, T. D.; Gurlo, A.; Clarke, D. R. *J. Mater. Chem.* **2012**, *22*, 4631–4636.
13
14
15 [57] Flahaut, D.; Funahashi, R.; Lee, K.; Ohta, H.; Koumoto, K. *25th International Con-*
16 *ference on Thermoelectrics*; 2006; pp 103–106.
17
18
19 [58] Hejtmanek, J.; Jirak, Z.; Marysko, M.; Martin, C.; Maignan, A.; Hervieu, M.;
20 Raveau, B. *Phys. Rev. B* **1999**, *60*, 14057–14065.
21
22
23 [59] Kobayashi, W.; Terasaki, I.; Mikami, M.; Funahashi, R.; Nomura, T.; Katsufuji, T. *J.*
24 *Appl. Phys.* **2004**, *95*, 6825–6827.
25
26
27 [60] Ohtaki, M.; Koga, H.; Tokunaga, T.; Eguchi, K.; Arai, H. *J. Solid State Chem.* **1995**,
28 *120*, 105–111.
29
30
31 [61] Kim, K. H.; Shim, S. H.; Shim, K. B.; Niihara, K.; Hojo, J. *J. Am. Ceram. Soc.* **2005**,
32 *88*, 628–632.
33
34
35 [62] Berardan, D.; Guilmeau, E.; Maignan, A.; Raveau, B. *Solid State Commun.* **2008**,
36 *146*, 97–101.
37
38
39 [63] Guilmeau, E.; Maignan, A.; Martin, C. *J. Electron. Mater.* **2009**, *38*, 1104–1108.
40
41
42 [64] Hebert, S.; Flahaut, D.; Martin, C.; Lemonnier, S.; Noudem, J.; Goupil, C.; Maig-
43 nan, A.; Hejtmanek, J. *Prog. Solid State Chem.* **2007**, *35*, 457–467.
44
45
46 [65] Wang, Y.; Lee, K. H.; Ohta, H.; Koumoto, K. *J. Appl. Phys.* **2009**, *105*, 103701.
47
48
49 [66] Muta, H.; Kurosaki, K.; Yamanaka, S. *J. Alloys Compd.* **2003**, *350*, 292–295.
50
51
52
53
54
55
56
57
58
59
60

- 1
2
3 [67] Park, K.; Ko, K. Y.; Seo, W. S.; Cho, W. S.; Kim, J. G.; Kim, J. Y. *J. Eur. Ceram. Soc.*
4 **2007**, *27*, 813–817.
5
6
7
8 [68] Sonne, M.; van Nong, N.; He, Z.; Pryds, N.; Linderoth, S. *8th European Conference*
9 *on Thermoelectrics*; 2010.
10
11
12
13 [69] Wang, N.; He, H. C.; Li, X. A.; Han, L.; Zhang, C. Q. *J. Alloys Compd.* **2010**, *506*,
14 293–296.
15
16
17
18 [70] Obara, H.; Yamamoto, A.; Lee, C. H.; Kobayashi, K.; Matsumoto, A.; Funahashi, R.
19 *Jpn. J. Appl. Phys., Part 2* **2004**, *43*, L540–L542.
20
21
22
23 [71] Muta, H.; Kurosaki, K.; Yamanaka, S. *J. Alloys Compd.* **2004**, *368*, 22–24.
24
25
26 [72] Jood, P.; Mehta, R. J.; Zhang, Y.; Peleckis, G.; Wang, X.; Siegel, R. W.; Borca-
27 Tasciuc, T.; Dou, S. X.; Ramanath, G. *Nano Lett.* **2011**, *11*, 4337–4342.
28
29
30
31 [73] Funahashi, R.; Shikano, M. *Appl. Phys. Lett.* **2002**, *81*, 1459–1461.
32
33
34 [74] Iwasaki, K.; Ito, T.; Nagasaki, T.; Arita, Y.; Yoshino, M.; Matsui, T. *J. Solid State*
35 *Chem.* **2008**, *181*, 3145–3150.
36
37
38
39 [75] Nan, J.; Wu, J.; Deng, Y.; Nan, C. W. *Solid State Commun.* **2002**, *124*, 243–246.
40
41
42 [76] Shikano, M.; Funahashi, R. *Appl. Phys. Lett.* **2003**, *82*, 1851–1853.
43
44
45 [77] Xu, G. J.; Funahashi, R.; Shikano, M.; Matsubara, I.; Zhou, Y. Q. *Appl. Phys. Lett.*
46 **2002**, *80*, 3760–3762.
47
48
49
50 [78] Shin, W.; Murayama, N. *Jpn. J. Appl. Phys., Part 2* **1999**, *38*, L1336–L1338.
51
52
53 [79] Zhou, Y. Q.; Matsubara, I.; Funahashi, R.; Sodeoka, S. *Mater. Lett.* **2001**, *51*, 347–
54 350.
55
56
57 [80] Yasukawa, M.; Kuniyoshi, S.; Kono, T. *Solid State Commun.* **2003**, *126*, 213–216.
58
59
60

- 1
2
3 [81] Yasukawa, M.; Murayama, N. *J. Mater. Sci. Lett.* **1997**, *16*, 1731–1734.
4
5
6 [82] Funahashi, R.; Mikami, M.; Urata, S.; Kouuchi, T.; Mizuno, K.; Chong, K. *MRS Proc.*
7
8 **2003**, *793*, S3.3.
9
10
11 [83] Sanmathi, C. S.; Retoux, R.; Noudem, J. G. *26th International Conference on Ther-*
12
13 *moelectrics*; 2007; pp 171–174.
14
15
16 [84] Choi, M. Y.; Kim, J. S. *Phys. Rev. B* **1999**, *59*, 192–194.
17
18
19 [85] Wang, C. X.; Chen, X. B.; Zhu, A. P. *Solid State Commun.* **2012**, *152*, 1067–1071.
20
21
22 [86] Takagi, H.; Batlogg, B.; Kao, H. L.; Kwo, J.; Cava, R. J.; Krajewski, J. J.; Peck, W. F.
23
24 *Phys. Rev. Lett.* **1992**, *69*, 2975–2978.
25
26
27 [87] Dong, X.; Wang, H.; Hua, Z.; Peng, S.; Dong, L.; Wang, Y. *J. Mater. Sci.: Mater.*
28
29 *Electron.* **2012**, *23*, 1210–1214.
30
31
32 [88] Kuriyama, H.; Nohara, M.; Sasagawa, T.; Takubo, K.; Mizokawa, T.; Kimura, K.;
33
34 Takagi, H. *25th International Conference on Thermoelectrics*; 2006; pp 97–98.
35
36
37 [89] Ono, Y.; Satoh, K.; Nozaki, T.; Kajitani, T. *25th International Conference on Thermo-*
38
39 *electrics*; 2006; pp 92–96.
40
41
42 [90] Tinh, N.; Tsuji, T. In *Physics and Engineering of New Materials*; Cat, D., Pucci, A.,
43
44 Wandelt, K., Eds.; Springer Berlin Heidelberg, 2009; Vol. 127; pp 209–217.
45
46
47 [91] Maignan, A.; Eyert, V.; Martin, C.; Kremer, S.; Fresard, R.; Pelloquin, D. *Phys. Rev.*
48
49 *B* **2009**, *80*, 115103.
50
51
52 [92] Muta, H.; Kurosaki, K.; Uno, M.; Yamanaka, S. *J. Alloys Compd.* **2002**, *335*, 200–
53
54 202.
55
56
57 [93] Kurosaki, K.; Kobayashi, H.; Yamanaka, S. *J. Alloys Compd.* **2003**, *350*, 340–343.
58
59
60

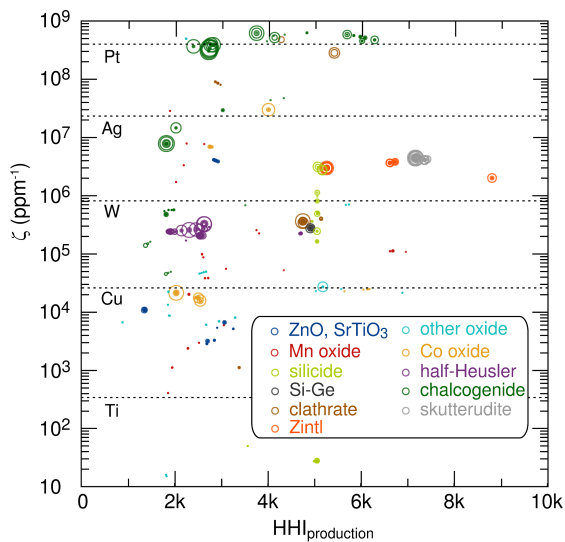
- 1
2
3 [94] Lee, S.; Wilke, R. H. T.; Trolier-McKinstry, S.; Zhang, S.; Randall, C. A. *Appl. Phys.*
4
5 *Lett.* **2010**, *96*, 031910.
6
7
8 [95] Lee, S.; Dursun, S.; Duran, C.; Randall, C. A. *J. Mater. Res.* **2011**, *26*, 26–30.
9
10
11 [96] Sakai, A.; Kanno, T.; Takahashi, K.; Omote, A.; Adachi, H.; Yamada, Y. *J. Am. Ceram.*
12
13 *Soc.* **2012**, *95*, 1750–1755.
14
15
16 [97] Joshi, G.; Lee, H.; Lan, Y.; Wang, X.; Zhu, G.; Wang, D.; Gould, R. W.; Cuff, D. C.;
17
18 Tang, M. Y.; Dresselhaus, M. S.; Chen, G.; Ren, Z. *Nano Letters* **2008**, *8*, 4670–4674.
19
20
21 [98] Saramat, A.; Svensson, G.; Palmqvist, A. E. C.; Stiewe, C.; Mueller, E.; Platzek, D.;
22
23 Williams, S. G. K.; Rowe, D. M.; Bryan, J. D.; Stucky, G. D. *J. Appl. Phys.* **2006**, *99*,
24
25 023708.
26
27
28 [99] Wolfing, B.; Kloc, C.; Teubner, J.; Bucher, E. *Phys. Rev. Lett.* **2001**, *86*, 4350–4353.
29
30
31 [100] Candolfi, C.; Aydemir, U.; Baitinger, M.; Oeschler, N.; Steglich, F.; Grin, Y. *J. Appl.*
32
33 *Phys.* **2012**, *111*, 043706.
34
35
36 [101] Nolas, G. S.; Cohn, J. L.; Slack, G. A.; Schujman, S. B. *Appl. Phys. Lett.* **1998**, *73*,
37
38 178–180.
39
40
41 [102] Kuznetsov, V. I.; Kuznetsova, L. A.; Kaliazin, A. E.; Rowe, D. M. *18th International*
42
43 *Conference on Thermoelectrics*; 1999; pp 177–180.
44
45
46 [103] Roudebush, J. H.; Toberer, E. S.; Hope, H.; Snyder, G. J.; Kauzlarich, S. M. *J. Solid*
47
48 *State Chem.* **2011**, *184*, 1176–1185.
49
50
51 [104] Katsuyama, S.; Matsushima, H.; Ito, M. *J. Alloys Compd.* **2004**, *385*, 232–237.
52
53
54 [105] Kimura, Y.; Tamura, Y.; Kita, T. *Appl. Phys. Lett.* **2008**, *92*, 012105.
55
56
57 [106] Muta, H.; Kanemitsu, T.; Kurosaki, K.; Yamanaka, S. *J. Alloys Compd.* **2009**, *469*,
58
59 50–55.
60

- 1
2
3 [107] Shutoh, N.; Sakurada, S. *J. Alloys Compd.* **2005**, *389*, 204–208.
4
5
6 [108] Fleurial, J. P.; Borshchevsky, A.; Caillat, T.; Morelli, D. T.; Meisner, G. P. *15th Inter-*
7
8 *national Conference on Thermoelectrics*; 1996; pp 91–95.
9
10
11 [109] Scherrer, H.; Scherrer, S. In *CRC Handbook of thermoelectrics*; Rowe, D. M., Ed.; CRC
12
13 Press: Boca Raton, FL, 1995.
14
15
16 [110] Skrabek, E. A.; Trimmer, D. S. In *CRC Handbook of thermoelectrics*; Rowe, D. M.,
17
18 Ed.; CRC Press: Boca Raton, FL, 1995.
19
20
21 [111] Chung, D. Y.; Choi, K. S.; Iordanidis, L.; Schindler, J. L.; Brazis, P. W.; Kan-
22
23 newurf, C. R.; Chen, B. X.; Hu, S. Q.; Uher, C.; Kanatzidis, M. G. *Chem. Mater.*
24
25 **1997**, *9*, 3060–3071.
26
27
28 [112] Gascoin, F.; Maignan, A. *Chem. Mater.* **2011**, *23*, 2510–2513.
29
30
31 [113] Kurosaki, K.; Kosuga, A.; Yamanaka, S. *J. Alloys Compd.* **2003**, *351*, 208–211.
32
33
34 [114] Ohta, M.; Yamamoto, A.; Obara, H. *J. Electron. Mater.* **2010**, *39*, 2117–2121.
35
36
37 [115] Wan, C. L.; Wang, Y. F.; Wang, N.; Koumoto, K. *Materials* **2010**, *3*, 2606–2617.
38
39
40 [116] Kanatzidis, M. G.; McCarthy, T. J.; Tanzer, T. A.; Chen, L. H.; Iordanidis, L.;
41
42 Hogan, T.; Kannewurf, C. R.; Uher, C.; Chen, B. X. *Chem. Mater.* **1996**, *8*, 1465–
43
44 1474.
45
46
47 [117] McGuire, M. A.; Reynolds, T. K.; DiSalvo, F. J. *Chem. Mater.* **2005**, *17*, 2875–2884.
48
49
50 [118] Kurosaki, K.; Kosuga, A.; Muta, H.; Uno, M.; Yamanaka, S. *Appl. Phys. Lett.* **2005**,
51
52 *87*, 061919.
53
54
55 [119] Sharp, J. W.; Sales, B. C.; Mandrus, D. G.; Chakoumakos, B. C. *Appl. Phys. Lett.*
56
57 **1999**, *74*, 3794–3796.
58
59
60

- 1
2
3 [120] Liu, H. L.; Shi, X.; Xu, F. F.; Zhang, L. L.; Zhang, W. Q.; Chen, L. D.; Li, Q.; Uher, C.;
4 Day, T.; Snyder, G. J. *Nat. Mater.* **2012**, *11*, 422–425.
5
6
7
8 [121] May, A. F.; Flage-Larsen, E.; Snyder, G. J. *Phys. Rev. B* **2010**, *81*, 125205.
9
10
11 [122] Raghavendra, N.; Gascoin, F.; Takahashi, H.; Hebert, S.; Guilmeau, E.; Terasaki, I.
12 Thermoelectric Properties of Hollandite Chalcogenides AM_5X_8 . Presented at EMRS
13 2012 Spring Meeting, Strasbourg, France, May 15–17.
14
15
16
17
18 [123] Mars, K.; Ihou-Mouko, H.; Pont, G.; Tobola, J.; Scherrer, H. *J. Electron. Mater.* **2009**,
19 *38*, 1360–1364.
20
21
22
23 [124] Zhang, Q.; Zhao, X. B.; Yin, H.; Zhu, T. J. *J. Alloys Compd.* **2008**, *464*, 9–12.
24
25
26 [125] Ito, M.; Tada, T.; Hara, S. *J. Alloys Compd.* **2006**, *408-412*, 363–367.
27
28
29 [126] Akasaka, M.; Iida, T.; Nemoto, T.; Soga, J.; Sato, J.; Makino, K.; Fukano, M.;
30 Takanashi, Y. *J. Cryst. Growth* **2007**, *304*, 196–201.
31
32
33
34 [127] Brown, S. R.; Kauzlarich, S. M.; Gascoin, F.; Snyder, G. J. *Chem. Mater.* **2006**, *18*,
35 1873–1877.
36
37
38
39 [128] Caillat, T.; Fleurial, J. P.; Borshchevsky, A. *J. Phys. Chem. Solids* **1997**, *58*, 1119–
40 1125.
41
42
43
44 [129] Kitagawa, H.; Noguchi, H.; Kiyabu, T.; Itoh, M.; Noda, Y. *J. Phys. Chem. Solids* **2004**,
45 *65*, 1223–1227.
46
47
48
49 [130] Toberer, E. S.; Rauwel, P.; Gariel, S.; Taftø, J.; Snyder, G. J. *J. Mater. Chem.* **2010**,
50 *20*, 9877–9885.
51
52
53
54 [131] Zevalkink, A.; Toberer, E. S.; Zeier, W. G.; Flage-Larsen, E.; Snyder, G. J. *Energy*
55 *Environ. Sci.* **2011**, *4*, 510–518.
56
57
58
59
60

1
2
3 [132] Toberer, E. S.; Zevalkink, A.; Crisosto, N.; Snyder, G. J. *Adv. Funct. Mater.* **2010**, *20*,
4
5 4375–4380.
6
7
8
9
10
11
12
13
14
15
16
17
18
19
20
21
22
23
24
25
26
27
28
29
30
31
32
33
34
35
36
37
38
39
40
41
42
43
44
45
46
47
48
49
50
51
52
53
54
55
56
57
58
59
60

TOC Graphics:



A data-driven review of thermoelectric materials, based around “mining” data from around 100 publications, and then devising an appropriate visualization scheme to view the mined data yields insights into performance metrics (indicated by the size of the circle) and economic realities.