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Weak antilocalization and conductance fluctuation in a single crystalline Bi nanowire

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We present the low temperature transport properties of an individual single-crystalline Bi nanowire grown by the on-film formation of nanowire method. The temperature dependent resistance and magnetoresistance of Bi nanowires were investigated. The phase coherence length was obtained from the fluctuation pattern of the magnetoresistance below 40 K using universal conductance fluctuation theory. The obtained temperature dependence of phase coherence length and the fluctuation amplitude indicates that the transport of electrons shows 2-dimensional characteristics originating from the surface states. The temperature dependence of the coherence length derived from the weak antilocalization effect using the Hikami–Larkin–Nagaoka model is consistent with that from the universal conductance fluctuations theory. © 2014 AIP Publishing LLC. [<http://dx.doi.org/10.1063/1.4863421>]

Semimetal bismuth has been widely employed for the study of electrical quantum effects, especially for the enhancement of thermoelectric performance due to quantum confinement.^{1–4} The low carrier effective mass, m^* , and long mean free path, l , of Bi result in high electrical mobility. For Bi nanowires (NW), the small carrier concentration, n , and the band overlap energy leads to semimetal-semiconductor (SMSC) transition as the diameter of the wires decreases.^{1,2} This SMSC transition is predicted for Bi nanowires with diameters less than 30 nm, depending on the crystal orientation, and is concurrent with the enhancement of the Seebeck coefficient.^{3,4} Studies of magnetic transport phenomenon to analyze the surface states of Bi nanowires show quantum oscillations, such as Shubnikov-de Hass (SdH) and Aharonov-Bohm (AB), superimposed on magnetoresistance (MR).^{5,6} These quantum oscillations in Bi nanowires suggest significant contribution from the 2-dimensional (2D) surface states in smaller diameter ($d < 50$ nm) Bi nanowires where the surface to volume ratio become drastically different from the bulk.^{7,8} Recent studies also revealed that the surface induced effect becomes more pronounced as the SMSC transition becomes apparent for the Bi nanowires with smaller diameters.⁹

Although the properties of low dimensional Bi nanowire, including low carrier concentration, long mean free path, low effective carrier mass, and anisotropic Fermi pockets, have been investigated, the quantitative analysis of properties related to the electron interference effects has not been reported in Bi nanowires with the diameters smaller than 35 nm, the regime where the SMSC transition and the surface states significantly change the electronic properties of nanowires. For a mesoscopic system with disorder and strong spin-orbit coupling, the phase coherent magneto-transport at low temperatures result in the appearance of weak antilocalization (WAL) and universal

conductance fluctuations (UCF) effects.^{10–12} These phenomena are affected by the phase-coherence length, l_ϕ , defined as the length where phase-coherent transport is preserved. Therefore, WAL and UCF are commonly used in determining the phase-coherence length of a material system.^{13–15}

In this paper, we report the WAL and UCF in a single-crystalline Bi nanowire grown by on-film formation of nanowire (OFF-ON) method.¹⁶ The phase coherence length of Bi nanowire was analyzed using magnetoresistance by the Hikami–Larkin–Nagaoka (HLN) model and UCF theory.^{17,18} The obtained temperature dependence of the phase coherence length indicates that the 2D transport behavior due to the surface states of Bi nanowire dominate the observed bulk transport characteristics.

Single-crystalline Bi nanowires were grown using OFF-ON method (Fig. 1(a)).¹⁶ Bi films with a 44 nm thickness were deposited on thermally oxidized Si (100) substrate using a custom-made sputter deposition system. Since the diameter of the nanowires is dependent on the grain size of the film, liquid nitrogen cooling system was used to obtain small diameter Bi nanowires close to 30 nm.¹⁹ The sputtering system was evacuated to 10^{-7} Torr prior to deposition, and the vacuum was maintained during sputtering under 2 mTorr Ar environment for 10 s. The growth rate of the Bi thin film was 44 Å/s. The Bi thin films were subsequently placed in a custom-made vacuum furnace for thermal annealing to grow nanowires by OFF-ON method (Fig. 1(b)). Annealing was performed in a vacuum of 10^{-7} Torr at 250 °C for 5 h. The growth direction of the nanowire is [100], which is vertical to the transmission electron microscope (TEM) image shown in Figure 1(c), and the electron diffraction (ED) pattern shows that the nanowire is a perfect single-crystalline (Fig. 1(d)).²⁰ A thermally oxidized Si (100) substrate was patterned with a grid of markers by photolithography and lift-off process before Bi nanowire dispersion. The Bi nanowires were dispersed by direct contact of grown nanowires to the Si

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FIG. 1. (a) A schematic representation of the growth of Bi nanowires by the OFF-ON method: Bi nanowires are grown to release thermal stress originating from thermal expansion mismatch between the SiO₂ and Bi thin film. (b) A scanning electron microscope (SEM) image of Bi nanowires grown on a Bi thin film. (c) A high-resolution TEM image of a typical Bi nanowire. The sample is sliced perpendicular to growth direction. (d) TEM electron diffraction pattern of a Bi nanowire. The [100] zone axis indicates the growth direction. (e) A SEM image of a typical 4-probe device based on the individual Bi nanowire geometry.

substrate. The combination of electron-beam lithography and a lift-off process was utilized to fabricate inner micron-scaled Cr/Au (2 nm/200 nm) electrodes. Plasma etching system was used to remove the oxide layer from the outer surface of the nanowires before the deposition of electrodes (Fig. 1(e)).^{21,22} The etching and deposition of electrodes were done *in-situ* without breaking the vacuum in order to prevent further formation of the oxide layer. All measurements were done in a helium cryostat equipped with a 9 T magnet at the temperature range from 1.85 K to 300 K. The magnetic field was perpendicular to the growth direction of the wires. The magnetoresistance was measured by a lock-in technique with an AC bias current of 100 nA. All measurements were done by the 4-terminal method in order to remove the contribution of contact resistance.

Figure 2(a) shows the temperature dependence of the resistance of three individual single-crystalline Bi nanowires

FIG. 2. (a) Temperature dependence of resistance for three different diameters 33, 39, and 45 nm at zero magnetic field. (b) Magnetoresistance as a function of magnetic field, normalized to the resistance at zero magnetic field, for the 33 nm diameter Bi nanowire.

with different diameters, 45, 39, and 33 nm, respectively. Since the carrier concentration of Bi ($\sim 3 \times 10^{17} \text{ cm}^{-3}$) is much smaller compared to normal metals, the temperature, T , dependent resistance, $R(T)$, measured in these nanowires are determined by the competition between mobility and carrier concentration.⁶ We observe $R(T)$ of the 45 nm Bi nanowire monotonically decreases with decreasing T . For thinner nanowires, $R(T)$ exhibits a non-monotonic temperature dependence, i.e., $R(T)$ first increases as T decreases followed by decreasing $R(T)$ for decreasing $T < 100$ K. Similar diameter dependent $R(T)$ behaviors were observed in Bi nanowires in a previous study,¹ where the non-monotonic dependence in small diameter nanowires is attributed to the competition between T dependent carrier-acoustic phonon scattering and the carrier non-specular surface scattering due to the classical size effect. In this study, we focus on the smallest diameter (33 nm) nanowire in the low temperature regime where the phase coherent quantum transport becomes apparent.

Figure 2(b) shows the magnetoresistance of individual Bi nanowire with 33 nm diameter at temperatures between 1.85 and 40 K, normalized to the resistance at zero magnetic field at the same temperature range. Magnetoresistance shows two physical phenomena related to the phase coherence length l_ϕ . The first is the narrow valley originating from WAL near the zero magnetic field.^{10,14} Usually, WAL occurs when the strong spin-orbit interaction reverses the sign of the resistance correction in the weak localization (WL) phenomena. Without a strong spin-orbit coupling, the WL effect

can be identified by the increase of resistance near the zero magnetic field because of the conductance correction by electron localization. In WAL, the additional phase shift arising from the spin-orbit interaction leads to the constructive interference of time-reversal pair of electron waves scattered by impurities, which in turn reduces the resistance around zero magnetic field. The second is the UCF which originates from the change of interference of the time-reversal pair of electron waves in a mesoscopic system, i.e., the size of system is not too different from l_ϕ , when the phase of the waves shifts according to the enclosed magnetic flux in the various closed loops of electron trajectories.^{11,15} These two effects are based on diffusive transport phenomena due to elastic scattering, and therefore can only be observed in low-temperature range where the inelastic scattering of phonons and electrons are frozen out. For Bi nanowires, the aforementioned effects were prominent below 10 K and were used to determine the phase coherence length, l_ϕ , of the material system.^{14,15,23,24}

For a more detailed analysis of the electron interference with UCF, the corresponding conductance fluctuations δG was separated from the magnetoresistance (Fig. 2(b)). The magnetoresistance was normalized to the conductance quantum e^2/h after the classical contribution G_0 was eliminated. The classical MR is proportional to the fundamental quantity $\omega_c \tau$, where ω_c and τ are the cyclotron frequency and relaxation time, respectively.⁵ The cyclotron frequency ω_c is proportional to the square of the magnetic field B .¹⁶ To extract the classical contribution, the quadratic fitting of measured MR was used. Figure 3(a) shows the conductance fluctuation at 1.85, 4, and 10 K, and the decrease of amplitude can be observed as the temperature increases. The amplitude of δG with a magnitude of one-tenth of e^2/h at 1.85 K agrees well with the correlation equation of the amplitude, δG ,

$$\delta G \propto (l_\phi/L)^{3/2}, \quad (1)$$

since the channel length, L , is relatively longer compared to the phase coherence length l_ϕ .¹¹

For the quantitative analysis of the magnitude of δG , root mean square (rms) of the conductance fluctuation, $\text{rms}(\delta G)$, which is defined by $\sqrt{\langle \delta G^2 \rangle}$, is used. Here, $\langle \dots \rangle$ expresses the ensemble average. Figure 3(b) shows the temperature dependence of $\text{rms}(\delta G)$ in log scale. The value increases as temperature decreases. The temperature dependence of the conductance fluctuation from 1.85 to 20 K is linear and fitted to be $T^{-0.77}$ as shown by the dash line. We will discuss the temperature dependence of $\text{rms}(\delta G)$ later after the analysis of the temperature dependence of the phase coherence length l_ϕ . Another important quantity that can be derived from δG is the correlation field B_c , which can be calculated through the autocorrelation function $F(\Delta B) = \langle \delta G(B + \Delta B) \delta G(B) \rangle$.¹⁸ B_c is defined as the full width at half maximum of the autocorrelation function $F(B_c) = \frac{1}{2} F(0)$. From the B_c value calculated from $\delta G(B)$, it can be seen that B_c linearly decreases proportional to $T^{0.43}$ as the temperature decreases from 40 K to 1.85 K (Fig. 3(c)). The temperature dependence of l_ϕ can be determined from the relation of B_c and the phase coherence length, l_ϕ , $l_\phi \approx \Phi_0/B_c d$, where Φ_0 and d are the magnetic flux quantum, $\Phi_0 = h/e$,^{21,25} and diameter, respectively.

FIG. 3. (a) Conductance fluctuations for 33 nm diameter as function of magnetic field at 1.85, 4, and 10 K. (b) Rms of the conductance fluctuations plotted against temperature. The temperature dependence of the conductance fluctuation is fitted to be $T^{-0.77}$, as shown by the dashed line. (c) Correlation field as a function of temperature. The dashed line represents the $T^{0.43}$ dependence.

A direct relation between B_c and l_ϕ allows us to find the temperature dependent phase coherence length. Figure 4(a) shows the obtained temperature dependence of the coherence length l_ϕ using UCF analysis, which increases with decreasing temperature proportional to $T^{-0.43}$. From the temperature dependence of l_ϕ , we can obtain the dimensionality of the system. Theoretically, the coherence length is proportional to $T^{-1/3}$, $T^{-1/2}$, and $T^{-3/4}$ for the one, two, and three-dimensional system, respectively.²⁶ Previous studies reported that Bi nanowire with the diameter of 100 nm has one-dimensional system¹² compared to three-dimensional bulk characteristics shown in 400 nm diameter.^{21,22} At the large diameter range, the 2D surface state effect is veiled due to the small surface to volume ratio and large bulk electrons compared to the surface carrier density.¹² However, the obtained exponent of -0.43 reflects that the system is a hybridization of one and two dimensions. The fact that this exponent is closer to the two-dimensional system value -0.5 suggests that the surface state of Bi plays a significant role in transport. Furthermore, the fitting which regards the saturation of l_ϕ at low temperature shows perfect linear dependence and the exponent of

FIG. 4. (a) Coherence lengths for the 33 nm diameter nanowire as a function of temperature obtained by the UCF theory (red) and the HNL model (blue). The black dashed lines show that the temperature dependence of the coherence lengths are fitted to be $T^{-0.43}$ and $T^{-0.52}$, respectively. The red dashed line represents the linear fitting result which regards the saturation of the coherence length at low temperatures. (b) Magneto-conductance around zero magnetic field shows a weak antilocalization effect. The colored curves present the fitting results by the HLN model. The y-axis scale is taken from the data obtained at 20 K. The other curves are incrementally shifted upward by e^2/h for clarity of presentation.

–0.51.¹⁵ In Bi thin films and Bi nanowires below 50 nm, surface state effect appears on bulk characteristics due to its large surface to volume ratio.^{7,8} The surface induced carrier concentration originating from the surface state was found to be $n_s = 5 \times 10^{12} \text{ cm}^{-2}$.⁹ The estimated carrier concentration of the 33 nm diameter nanowire was calculated to be $6.0 \times 10^{18} \text{ cm}^{-3}$, based on the relation of the nanowire carrier concentration $2n_s/r$, where r is the radius of the nanowire. The surface induced carrier concentrations of the nanowires are significantly large relative to that of the one-dimensional system, and, as a result, the system is dominated by two-dimensional characteristics. The quantum transport parameters with variation of temperature are listed in Table I.

We now discuss the temperature dependence of $\text{rms}(\delta G) \sim T^{-0.77}$ observed above. Since the estimated l_ϕ in all temperature range of our experiment is less than 500 nm, which is much less than the channel length $L = 1 \mu\text{m}$ in our device, we expect Eq. (1) to hold for the amplitude fluctuation for δG . From the previous discussion, we can approximate $l_\phi \sim T^{-1/2}$, therefore, we expect $\text{rms}(\delta G) \sim l_\phi^{3/2} \sim T^{-3/4}$. The temperature dependence of $\text{rms}(\delta G)$ observed in the Bi NW is proportional to $T^{-0.77}$, which is close to the theoretically expected value from the UCF theory,¹⁸ and reaffirms

TABLE I. Quantum transport properties obtained from UCF and WAL effects in 33 nm Bi nanowire.

T (K)	UCF rms (δG) (e^2/h)	UCF B_c (T)	UCF l_ϕ (nm)	WAL l_ϕ (nm)
1.85	0.050	0.475	264.0	214.0
4	0.038	0.528	237.4	186.5
10	0.009	0.822	152.5	109.5
20	0.009	1.149	109.1	62.9
40	0.004	1.712	73.2	

that the dimensionality of the system is dominated by 2D characteristics.

The WAL of MR in weak magnetic field range (–1.4 to 1.4 T) is similar to the UCF. Since WAL is sensitive to the phase coherence, the temperature dependence of l_ϕ can be estimated from the background of magneto-conductance change, $\delta G_{\text{WAL}}(B) = G(B) - G(0)$. In the HLN theory,¹⁷ δG_{WAL} can be related to l_ϕ via the phase coherence magnetic field $B_\phi = \frac{\hbar}{4el_\phi^2}$, using the equation

$$\delta G_{\text{WAL}}(B) = -\alpha \frac{e^2}{2\pi^2\hbar} \left[\psi\left(\frac{1}{2} + \frac{B_\phi}{B}\right) - \ln\left(\frac{B_\phi}{B}\right) \right], \quad (2)$$

where α and ψ are the fitting parameter and the digamma function, respectively. Fitting Eq. (2) for B_ϕ for the curves of $\delta G_{\text{WAL}}(B)$ at different temperature (Fig. 4(b)), we can then find the temperature dependent l_ϕ . Figure 4(a) shows l_ϕ obtained from the WAL analysis compared to the previously obtained l_ϕ from the UCF analysis. We find that l_ϕ from the WAL analysis shows $T^{-0.52}$ dependence, which is consistent with that obtained from the UCF theory.

In summary, the magnetoresistance of an individual single-crystalline Bi nanowire grown by the OFF-ON method was measured. The observed dip near zero magnetic field and the fluctuation with magnetic field sweep originate from WAL effect and UCF, respectively. The temperature dependence of the phase coherence length obtained from the UCF theory confirmed that the transport of carriers shows 2D characteristics that originates from the surface states of Bi nanowires. The exponential coefficient from the temperature dependence of the coherence length derived from the WAL effect using HLN model is consistent with the value obtained from the UCF theory. Our investigations demonstrate that the surface states of Bi nanowires dominate the bulk transport characteristics in the diameter range close to 30 nm.

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¹J. Heremans, C. M. Thrush, Y. M. Lin, S. Cronin, Z. Zhang, M. S. Dresselhaus, and J. F. Mansfield, *Phys. Rev. B* **61**, 2921–2930 (2000).

²Z. Zhang, X. Sun, M. S. Dresselhaus, J. Y. Ying, and J. Heremans, *Phys. Rev. B* **61**, 4850–4861 (2000).

³L. D. Hicks and M. S. Dresselhaus, *Phys. Rev. B* **47**, 16631–16634 (1993).

- ⁴Y. M. Lin, X. Sun, and M. S. Dresselhaus, *Phys. Rev. B* **62**, 4610–4623 (2000).
- ⁵F. Y. Yang, K. Liu, K. Hong, D. H. Reich, P. C. Searson, and C. L. Chien, *Science* **284**, 1335–1337 (1999).
- ⁶T. E. Huber, A. Nikolaeva, D. Gitsu, L. Konopko, C. A. Foss, Jr., and M. J. Graf, *Appl. Phys. Lett.* **84**, 1326–1328 (2004).
- ⁷A. Nikolaeva, D. Gitsu, L. Konopko, M. J. Graf, and T. E. Huber, *Phys. Rev. B* **77**, 075332 (2008).
- ⁸P. Hofmann, *Prog. Surf. Sci.* **81**, 191–245 (2006).
- ⁹T. E. Huber, A. Adeyeye, A. Nikolaeva, L. Konopko, R. C. Johnson, and M. J. Graf, *Phys. Rev. B* **83**, 235414 (2011).
- ¹⁰C. W. J. Beenakker and H. van Houten, *Solid State Physics: Advances in Research and Applications* (Academic Press, New York, 1992), vol. 44, p. 1.
- ¹¹H. Ibach and H. Lüth, *Solid State Physics: An Introduction to Principles of Materials Science*, (Springer, New York, 2009), p. 278.
- ¹²N. Marcano, S. Sangiao, M. Plaza, A. F. Pacheco, R. Cordoba, M. C. Sanchez, L. Morellon, M. R. Ibarra, and J. M. De Teresa, *Appl. Phys. Lett.* **96**, 082110 (2010).
- ¹³Z. Li, T. Chen, H. Pan, F. Song, B. Wang, J. Han, Y. Qin, X. Wang, R. Zhang, J. Wan, D. Xing, and G. Wang, *Sci. Rep.* **2**, 595 (2012).
- ¹⁴S. Matsuo, T. Koyama, K. Shimamura, T. Arakawa, Y. Nishihara, D. Chiba, K. Kobayashi, and T. Ono, *Phys. Rev. B* **85**, 075440 (2012).
- ¹⁵Ch. Blömers, Th. Schäpers, T. Richter, T. Calarco, H. Lüth, and M. Marso, *Appl. Phys. Lett.* **92**, 132101 (2008).
- ¹⁶W. Shim, J. Ham, K. Lee, W. Jeung, M. Johnson, and W. Lee, *Nano Lett.* **9**, 18–22 (2009).
- ¹⁷S. Hikami, A. Larkin, and Y. Nagaoka, *Prog. Theor. Phys.* **63**, 707 (1980).
- ¹⁸P. Lee, A. Stone, and H. Fukuyama, *Phys. Rev. B* **35**, 1039–1070 (1987).
- ¹⁹S. Lee, J. Ham, K. Jeon, J. Noh, and W. Lee, *Nanotechnology* **21**, 405701 (2010).
- ²⁰J. Roh, K. Hippalgaonkar, J. Ham, R. Chen, M. Li, P. Ercius, A. Majumdar, W. Kim, and W. Lee, *ACS Nano* **5**, 3954–3960 (2011).
- ²¹W. Shim, J. Ham, J. Kim, and W. Lee, *Appl. Phys. Lett.* **95**, 232107 (2009).
- ²²K. Lee, S. Lee, S. N. Holmes, J. Ham, W. Lee, and C. H. W. Barnes, *Phys. Rev. B* **82**, 245310 (2010).
- ²³J. Chen, H. J. Qin, F. Yang, J. Liu, T. Guan, F. M. Qu, G. H. Zhang, J. R. Shi, X. C. Xie, C. L. Yang, K. H. Wu, Y. Q. Li, and L. Lu, *Phys. Rev. Lett.* **105**, 176602 (2010).
- ²⁴H. T. He, G. Wang, T. Zhang, I. K. Sou, G. K. L. Wong, J. N. Wang, H. Z. Lu, S. Q. Shen, and F. C. Zhang, *Phys. Rev. Lett.* **106**, 166805 (2011).
- ²⁵C. W. J. Beenakker and H. Houten, *Phys. Rev. B* **37**, 6544–6546 (1988).
- ²⁶B. L. Altshuler, A. G. Aronov, and D. E. Khmel'nitskii, *J. Phys. C: Solid State Phys.* **15**, 7367–7386 (1982).