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Atomic Layer Deposition of Zn(O,S) Thin Films with Tunable Electrical Properties by Oxygen Annealing

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Abstract

Zinc oxysulfide, Zn(O,S), films grown by atomic layer deposition (ALD) were annealed in oxygen to adjust the carrier concentration. The electron carrier concentration of Zn(O,S) can be reduced by several orders of magnitude from \(10^{19}\) to \(10^{15}\) cm\(^{-3}\) by post-deposition annealing in oxygen at temperatures from 200°C to 290°C. In the case of Zn(O,S) with S/Zn = 0.37, despite the considerable change in the electron carrier concentration, the bandgap energy decreased by only \(\sim 0.1\) eV, and the crystallinity did not change much after annealing. The oxygen/zinc ratio increased by 0.05 after annealing, but the stoichiometry remained uniform throughout the film.

Keywords: Zinc Oxysulfide, Buffer Layer, Atomic Layer Deposition, Thin-Film Solar Cells

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Cu(In,Ga)(S,Se)\(_2\) (CIGS) is one of the most reliable materials used in thin-film solar cells, but currently, the most efficient CIGS-based solar cells use CdS\(^1\),\(^2\) a toxic material, as an \(n\)-type buffer layer between the \(p\)-type CIGS absorber layer and the ZnO/transparent conducting oxide (TCO) layers. Buffer layers are critical in reducing interface recombination and giving optimum band alignment across the junction.\(^3\),\(^4\) It is therefore of interest to study alternative materials that better satisfy the earth-abundance and non-toxicity requirements for low-cost and environmentally compatible large-scale production. Previous research has shown that zinc oxysulfide—Zn(O,S)—grown by atomic layer deposition (ALD) is a promising alternative to CdS in CIGS solar cells.\(^5\),\(^6\)

The larger bandgap of Zn(O,S) (\(E_g \approx 2.6 \text{ to } 3.8\) eV) compared to the bandgap of CdS (\(E_g \approx 2.4\) eV) reduces photocurrent loss in the short-wavelength region.\(^7\) In addition, the bandgap of Zn(O,S) and the conduction band offset at the buffer/absorber interface can be finely tuned by altering the stoichiometry of Zn(O,S).\(^3\),\(^6\),\(^8\),\(^9\) Various growth methods have been previously reported for Zn(O,S), such as sputtering,\(^8\),\(^10\),\(^11\) chemical bath deposition (CBD),\(^12\) pulse laser deposition (PLD),\(^13\),\(^14\) and ALD.\(^3\),\(^6\),\(^15\),\(^16\) ALD has the advantage of easily controlling the stoichiometry of multicomponent films by simply tailoring the pulse ratio of the precursors.

In addition to the progress of replacing CdS with Zn(O,S) buffer layers in CIGS-based solar cells, much research has also been motivated to replace the expensive indium-based CIGS absorber layer with more earth-abundant materials, such as Cu\(_2\)ZnSn(Se,S)\(_4\)\(^17\),\(^19\) and SnS.\(^20\),\(^22\)

These new absorber layers need an \(n\)-type material partner such as Zn(O,S) to serve as a buffer layer. In fact, Zn(O,S) buffer layers have already been integrated with SnS absorber layers to produce solar cells with record efficiency for SnS-based solar cells.\(^23\) With increasing interest in such research areas, it is important to have a better understanding of ALD Zn(O,S) and the tunability of its properties to further improve the performance of thin-film solar cells using earth-abundant and non-toxic elements.

The tunability of the electrical properties of Zn(O,S) is important since the \(n\)-type buffer layer will affect the recombination of free carriers at the absorber-buffer interface.\(^24\) An optimum of carrier concentration for Zn(O,S) is needed since the carrier concentration will also affect the Fermi level within the buffer layer. In order to achieve high efficiency, the buffer layer carriers must be depleted from the absorber region to reduce recombination at the interface.

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Cu(In,Ga)(S,Se)\(_2\) (CIGS) is one of the most reliable materials used in thin-film solar cells.
carrier concentration as-deposited and oxygen annealed films and 0.42, respectively, as determined by RBS analysis. Hall measurements were over temperature shows used to estimate the elemental distribution of O, S, Zn, Si, and C Rutherford backscattering spectroscopy (RBS) showed that the O/Zn ratio increased by 0.05 after oxygen annealing while the S/Zn ratio remained almost the same. Depth profiling by XPS was used to estimate the elemental distribution of O, S, Zn, Si, and C throughout the film. Figure 1 shows a uniform distribution of elements in Zn(O,S) with S/Zn = 0.37 annealed at the highest temperature (290°C). RBS also indicated uniform stoichiometry throughout the film after oxygen annealing. These results imply that the diffusion of oxygen took place throughout the film and was not confined to the surface, which is also reflected in the Hall data of Fig. 2.

The dependence of oxygen anneal temperature on the electrical properties of Zn(O,S) is shown as a function of the ratio of the sulfur content S/Zn for different oxygen anneal temperatures in Fig. 2. For the as-deposited films, Zn(O,S) 19:1, 9:1, 7:1, 6:1, 5:1, and 4:1 have S/Zn ratios of 0.09, 0.26, 0.37, 0.50, 0.64, and 0.73, respectively, and O/Zn ratios of 0.99, 0.82, 0.72, 0.64, 0.50, and 0.42, respectively, as determined by RBS analysis. Hall measurements were over the detection limit for films with high sulfur content such as Zn(O,S) with S/Zn = 0.50, 0.64, and 0.73 due to their high resistivities. Overall, the resistivity had a tendency to increase with sulfur content for the as-deposited and oxygen annealed films (Fig. 2a). For the as-deposited Zn(O,S) films, the electron carrier concentration remained around the \(10^{19} \text{ cm}^{-3} \) range despite the increase of sulfur content (Fig. 2b). Annealing the Z(O,S) films in O or N has reduced the sulfur content, as indicated by RBS analysis. However, further studies on the impact of the carrier concentration of the buffer layer on the solar cell performance are still needed. Previously, it has been shown that post-deposition annealing in oxygen can help to reduce oxygen defects, such as oxygen vacancies. Oxygen vacancies contribute to the high carrier concentration in various oxides. In this paper, we report that Zn(O,S) can be annealed in oxygen to reduce its electron carrier concentration by up to four orders of magnitude, with negligible influence on the bandgap and crystallinity of the potential buffer layer material.
Electron carrier mobility showed a tendency to decrease with increasing sulfur content for the as-deposited films, which may be due to increased scattering from the disorder introduced by random substitution of sulfur for oxygen. However, it was difficult to notice clear trends with sulfur content in the mobility of annealed films. The mobilities of the oxygen annealed films (0.8 – 7.6 cm²/V·s) tended to be lower than the as-deposited films (27.5 – 27.8 cm²/V·s), for the films with lower sulfur content (Fig. 2c). Hall measurements of mobility were not possible for the highly resistive Zn(O,S) films with S/Zn = 0.37 annealed at 290°C.

Stability of the Zn(O,S) films was investigated by repeating the Hall measurements of films that were kept under atmosphere condition for approximately one year, as shown in Fig. 2. Carrier concentrations increased by approximately an order of magnitude after ~1 yr under atmosphere condition for films with lower carrier concentrations. This may be due to the increased hydroxyl groups within the film over time, since ZnO-based films are known for easily picking up moisture from the atmosphere.29 The increase in carrier concentration over time was much less significant for films with higher carrier concentrations. There is no clear trend for the mobilities of the films.

Figure 3(a) shows a plot of α² vs. photon energy for the as-deposited ZnO and Zn(O,S) with S/Zn = 0.37, and Zn(O,S) with S/Zn = 0.37 annealed in O₂ at temperatures between 200°C and 290°C. Bandgap energy was then estimated using Tauc’s relation for direct transitions:30

\[ \alpha(h\nu) \propto (h\nu - E_g)^{1/2} \]  

where \( \alpha(h\nu) \) is the absorption coefficient, \( h\nu \) is the photon energy, and \( E_g \) is the optical bandgap, assuming that the electron and hole effective masses are constant. In Fig. 3(a), the steeper slope for the as-deposited ZnO compared to the Zn(O,S) films with S/Zn = 0.37 can be qualitatively explained by the disorder-induced band-tail states of Zn(O,S). For Zn(O,S) films with S/Zn = 0.37, the bandgap energy decreased by ~0.1 eV after the film was annealed in O₂ at 200°C, but higher annealing temperatures did not have any significant additional effect on the bandgap energy, as shown in Fig. 3(b).

X-ray diffraction of the as-deposited ZnO and Zn(O,S) with S/Zn = 0.37, and Zn(O,S) with S/Zn = 0.37 annealed in O₂ at different temperatures grown on quartz substrates are shown in Fig. 4. For the as-deposited films, once sulfur was added, the films remained polycrystalline, but the peak intensities decreased. This reduction may be due to an amorphous component of the Zn(O,S) films, which does not contribute to the diffraction peaks.

The lattice constant, vertical grain size, and nonuniform distribution of local strain in the films were determined from diffraction. To separate the \( K\alpha_{1} \) and \( K\alpha_{2} \) peaks, double-peak Lorentzian functions were used, and the instrumental peak broadening was taken into account.31-33 Nonuniform distribution of local strain and grain size were then estimated from the Scherrer equation of ∆k vs. k (scattering vector \( k = (4\pi/\lambda)\sin\theta \)):

\[ \Delta k = \frac{\Delta d}{d} k = \frac{2\pi}{D} \]  

where \( \Delta k \) is the full width at half maximum (FWHM), \( \Delta d/d \) is the nonuniform distribution of local strain, and \( D \) is the grain size. Substitution of sulfur increases the lattice constant of Zn(O,S) \( (a = 0.336 \text{ nm}) \) over that of ZnO \( (a = 0.323 \text{ nm}) \), while Vegard’s law predicts \( a = 0.374 \text{ nm} \). Oxygen annealing does not alter the lattice constant of Zn(O,S) significantly (Fig. 5a). No significant difference in grain size was found between ZnO and Zn(O,S) (Fig. 5b). This result is not consistent with previous studies that observed grain size reduction for intermediate compositions of Zn(O,S).24 Although the oxygen anneal temperature had a large effect on the electrical properties of Zn(O,S) with S/Zn = 0.37, the anneal temperature did not produce any noticeable modification of the grain size and local strain, as shown in Figs. 5b and 5c. XRD analysis of Zn(O,S) films grown on quartz substrates show very similar grain size and texture to films grown on Si, as shown in Figs. S1 and S2 (see Ref. 34), which implies that the crystallization behavior of the films is independent of the substrate type.

The low dependence of oxygen anneal temperature on the vertical and lateral grain sizes
was confirmed by the cross-sectional and plan-view FESEM images shown in Fig. 6. Although the addition of sulfur to ZnO changed the shape and size of the lateral grains, it was difficult to observe any noteworthy change in the vertical and lateral grains of Zn(O,S) with S/Zn = 0.37 by oxygen annealing.

In conclusion, it was demonstrated that an anneal temperature range of only 200°C – 290°C reduced the electron carrier concentration of ALD Zn(O,S) by four orders of magnitude from $10^{19}$ to $10^{15}$ cm$^{-3}$. The electrical properties of Zn(O,S) are strongly affected by the sulfur content and the oxygen annealing temperature. Bandgaps of the Zn(O,S) films with S/Zn = 0.37 were shown to change by only ~0.1 eV with O$_2$ annealing. Although electrical properties were modified to a large extent by oxygen annealing, the annealing temperature investigated in this study had an insignificant effect on the bandgap and crystallinity of Zn(O,S) films with S/Zn = 0.37. RBS showed that oxygen was added to the films by annealing, but the sulfur content remained unchanged. Depth profiling by XPS showed that the distribution of elements through the film remained uniform after annealing in oxygen.

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References

Figure Captions

Fig. 1. Depth profile of Zn(O,S) with S/Zn = 0.37, up to the Si substrate.

Fig. 2. Plots of (a) resistivity, (b) electron carrier concentration, and (c) electron mobility vs. S/Zn, for the as-deposited Zn(O,S) and Zn(O,S) samples annealed at 200°C, 230°C, 260°C, and 290°C in O_2. Measurements were repeated for films after ~1 yr with S/Zn = 0.37, as shown on the right side of the plots.

Fig. 3. Plots of (a) α^2 vs. hν and (b) bandgap energy vs. O_2 annealing temperature for Zn(O,S) with S/Zn = 0.37.

Fig. 4. X-ray diffraction for the as-deposited ZnO, as-deposited Zn(O,S) with S/Zn = 0.37, and Zn(O,S) with S/Zn = 0.37, oxygen annealed at 200°C, 230°C, 260°C, and 290°C on quartz substrates.

Fig. 5. Plots of (a) lattice constant, (b) grain size, and (c) local strain vs. O_2 annealing temperature for the as-deposited ZnO, as-deposited Zn(O,S) with S/Zn = 0.37, and Zn(O,S) with S/Zn = 0.37, oxygen annealed at different temperatures on quartz substrates.

Fig. 6. Cross-sectional and plan-view FESEM images for the as-deposited ZnO, as-deposited Zn(O,S) with S/Zn = 0.37, and Zn(O,S) with S/Zn = 0.37, oxygen annealed at 200°C, 230°C, 260°C, and 290°C.
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Fig. S1. X-ray diffraction for the as-deposited ZnO, as-deposited Zn(O,S) with \(S/Zn = 0.37\), and Zn(O,S) with \(S/Zn = 0.37\), annealed in \(O_2\) at various temperatures on Si substrates.
Fig. S2. Plots of (a) lattice constant, (b) grain size, and (c) local strain vs. O$_2$ annealing temperature for the as-deposited ZnO, as-deposited Zn(O,S) with S/Zn = 0.37, and Zn(O,S) with S/Zn = 0.37, oxygen annealed at different temperatures on Si substrates.